## Gaussian Processes for Force Fields

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EPSRC
Engineering and Physical Sciences Research Council

## Part I

# Bayesian inference and Gaussian process regression 

Aldo Glielmo<br>Physics Department, King's College London

## The aim of classical potentials



## References:

Skinnert et al. Mod. \& Sim. in Materials (1994 )
Behler, J., \& Parrinello, M. PRL (2007)
Bartók et al. PRL (20।0)
Li, Z., Kermode, J. R., \& De Vita, A. PRL (2015)

## The aim of classical potentials



Systematic approaches to implicitly model the electronic degrees of freedom

References:

$$
E=E\left(\left\{\mathbf{r}_{i}\right\} \mid \varepsilon_{0}\right)
$$

Skinnert et al. Mod. \& Sim. in Materials (1994 )
Behler, J., \& Parrinello, M. PRL (2007)
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## Learning force fields: problem setup

- Quantum system of $N$ atoms
at a specific time

- The computational cost of calculating energies and forces scales badly with N and also has high prefactors


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Input Space:

- Configuration $\boldsymbol{\rho}$
- $3 x M$ dimensional
- typically M~ 30-40
- The computational cost of calculating energies and forces scales badly with N and also has high prefactors



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Output space:

- Local energy $\boldsymbol{\varepsilon}$ of central atom
- The force $\mathbf{f}$ is obtained through differentiation
$\stackrel{\text { f }}{ } \downarrow$
Force


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Output space:

- Local energy $\varepsilon$ of central atom
- The force $\mathbf{f}$ is obtained through differentiation

f $\downarrow$
Force
- The function $\mathbf{f}$ to be learned gives the force as a function of the local configuration $\rho$ around a central atom
- The existence of such map can be assumed in most physical systems
- Linear scaling with $N$ is guaranteed by the locality of the representation


## Traditional approach

- The traditional way of fitting potential energies and forces involves a lengthy trial and error procedure of careful selection of meaningful parametric functional forms


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A Lennard-Jones parametric potential

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Table 1


Optimized values of fitting parameters of the EAM potentials for $\mathrm{Ni}, \mathrm{Al}$ and $\mathrm{Ni}_{3} \mathrm{Al}$

| A1 |  | Ni |  | $\mathrm{Ni}_{3} \mathrm{Al}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Parameter | Value | Parameter | Value | Parameter | Value |
| $r_{\mathrm{c}}(\mathrm{nm})$ | 0.6725 | $r_{\mathrm{c}}(\mathrm{nm})$ | 0.5168 | $r_{\mathrm{c}}(\mathrm{nm})$ | 0.6500 |
| $h_{\text {c }}(\mathrm{nm})$ | 0.3294 | $h_{\text {c }}(\mathrm{nm})$ | 0.3323 | $h_{\mathrm{c}}(\mathrm{nm})$ | 0.2658 |
| $V_{0}(\mathrm{eV})$ | $-3.5032 \times 10^{3}$ | $V_{0}(\mathrm{eV})$ | $-3.5126 \times 10^{3}$ | $V_{0}(\mathrm{eV})$ | 0.6068 |
| $r_{1}(\mathrm{~nm})$ | 0.2858 | $r_{1}(\mathrm{~nm})$ | $3.8673 \times 10^{-5}$ | $r_{1}(\mathrm{~nm})$ | 0.4834 |
| $b_{1}$ | $8.5951 \times 10^{-2}$ | $b_{1}$ | $4.7067 \times 10^{-3}$ | $b_{1}$ | 2.9013 |
| $b_{2}$ | $5.0124 \times 10^{-2}$ | $b_{2}$ | 0.15106 | $b_{2}$ | 1.0001 |
| $\delta(\mathrm{eV})$ | $3.7503 \times 10^{3}$ | $\delta(\mathrm{eV})$ | $3.6046 \times 10^{3}$ | $\delta(\mathrm{eV})$ | -3.4108 |
|  | $2.0080 \times 10^{1}$ | $y$ | $1.9251 \times 10^{1}$ |  |  |
| $\gamma(1 / \mathrm{nm})$ | $4.2799 \times 10^{1}$ | $\gamma(1 / \mathrm{nm})$ | $1.6802 \times 10^{3}$ | $g_{\mathrm{Ni}}(\mathrm{eV})$ | $5.8549 \times 10^{1}$ |
| $B_{0}(\mathrm{~nm})$ | $1.1927 \times 10^{4}$ | $B_{0}(\mathrm{~nm})$ | $1.1914 \times 10^{4}$ | $g_{\text {Al }}(\mathrm{eV})$ | $-1.8162 \times 10^{1}$ |
| $C_{0}\left(1 / \mathrm{nm}^{3}\right)$ | $8.6029 \times 10^{1}$ | $C_{0}\left(1 / \mathrm{nm}^{3}\right)$ | $2.0329 \times 10^{2}$ |  |  |
| $r_{0}(\mathrm{~nm})$ | $5.2755 \times 10^{-2}$ | $r_{0}(\mathrm{~nm})$ | $-0.3138$ |  |  |
| $\beta$ | $0.4890 \times 10^{-2}$ | $\beta$ | $0.4890 \times 10^{-2}$ |  |  |

Typical table of parameters of an EAM potential (from Mishin, Acta Materiali (2004))

## The "new" data-driven approach



- Very flexible models able to capture much more information from data


## The "new" data-driven approach

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Generalized Neural-Network Representation of High-Dimensional Potential-Energy Surfaces

Jörg Behler and Michele Parrinello
Phys. Rev. Lett. 98, 146401 - Published 2 April 2007
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Featured in Physics
Gaussian Approximation Potentials: The Accuracy of Quantum Mechanics, without the Electrons
Albert P. Bartók, Mike C. Payne, Risi Kondor, and Gábor Csányi
Phys. Rev. Lett. 104, 136403 - Published 1 April 2010
1994
IOPScience Journals Books Publishing Support Login
Modelling and Simulation in Materials Science and Engineering

Neural networks in computational materials science:
training algorithms
A J Skinner and J Q Broughton

"Our long term goal in applying neural networks is to demonstrate that they can be used to interpolate complex and expensive $a b$ initio databases that describe interactions between atoms."

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## Bayesian inference 101

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Dataset (e.g. configurations and forces from a DFT calculation)
$p(f)$
Prior: Our beliefs about the function before we observe any data
(e.g., a Lennard Jones with Gaussian distributed parameters)

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Posterior


## Probabilistic interpretation of least squares

> Bayes' Theorem
> $p(\mathbf{w} \mid \mathscr{D}) \propto \frac{p(\mathscr{D} \mid \mathbf{w}) p(\mathbf{w})}{p(\mathscr{D})}$

## Probabilistic interpretation of least squares

## Bayes' Theorem

$p(\mathbf{w} \mid \mathscr{D}) \propto \frac{p(\mathscr{D} \mid \mathbf{w}) p(\mathbf{w})}{p(\mathscr{D})}$
Regularised squared error loss

$$
L(\mathbf{w})=\sum_{i}\left\|f_{i}-f_{\mathbf{w}}\left(\rho_{i}\right)\right\|^{2}+\alpha\|\mathbf{w}\|^{2}
$$



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Gaussian prior and Gaussian likelihood
$p(\mathbf{w}) \propto e^{-\|\mathbf{w}\|^{2} / 2 \sigma_{\mathbf{w}}^{2}}$
$p\left(f_{i} \mid \mathbf{w}\right) \propto e^{-\left\|f_{i}-f_{\mathbf{w}}\left(\rho_{i}\right)\right\|^{2} / 2 \sigma_{f}^{2}}$

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$p\left(f_{i} \mid \mathbf{w}\right) \propto e^{-\left\|f_{i}-f_{\mathbf{w}}\left(\rho_{i}\right)\right\|^{2} / 2 \sigma_{f}^{2}}$


$$
-\ln p(\mathbf{w} \mid \mathscr{D}) \propto \sum_{i}\left\|f_{i}-f_{\mathbf{w}}\left(\rho_{i}\right)\right\|^{2}+\frac{\sigma_{f}^{2}}{\sigma_{\mathbf{w}}^{2}}\|\mathbf{w}\|^{2}
$$

Minimising a regularised squared loss corresponds to maximising a posterior distribution!


Prior

[Rasmussen and Williams 2006]

Posterior


Prior information

$$
p(f \mid \mathscr{D}) \propto p(\mathscr{D} \mid f) p(f) \mid
$$

Likelihood
$\mathscr{D}=(\rho, \mathbf{f})_{1},(\rho, \mathbf{f})_{2}$,
Traditional approaches:

- High prior information (parametric functional forms)
- Hence:

X limited in their accuracy if prior is wrong
$\checkmark$ simple to interpret
$\checkmark$ very fast
$\checkmark$ tend to be more transferable


## New ML approaches:

- Low prior information (Neural Networks, non parametric regression)
- Hence:
$\checkmark$ very flexible (potentially very accurate)
X difficult to interpret
$X$ relatively slow
X transferability can be problematic

For robustness, for interpretability, and since we will never have infinite data:
We need a way to select the appropriate model complexity

## Gaussian process prior

A Gaussian process can be used as prior distribution $p(f(\rho))=\mathscr{G} \mathscr{P}\left(0, k\left(x, x^{\prime}\right)\right)$


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$p(f(\rho))=\mathscr{G} \mathscr{P}\left(0, k\left(x, x^{\prime}\right)\right)$

Using a Gaussian likelihood, the posterior Gaussian process can be found analytically
$p(f(x) \mid \mathscr{D})=\mathscr{G} \mathscr{P}\left(\hat{f}(x), \hat{k}\left(x, x^{\prime}\right)\right)$
$\hat{f}(x)=\sum_{i=1}^{N} k\left(x, x_{i}\right) \alpha_{i} \quad$ Predictions
$\hat{\sigma}^{2}(x)=\hat{k}(x, x)$
Predicted variance (measure of uncertainty)


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- Very flexible regression model (provably equivalent to a NN with a single, infinite hidden layer)
- Principled uncertainty predictions (crucial for transferability and validation)
- Prior knowledge can be included into the kernel function


## GP visualisation

Squared exponential kernel with length scale $\ell$
$k(x, x)=e^{-\left(x-x^{\prime}\right)^{2} / 2 \ell^{2}}$

Gaussian likelihood with "noise" parameter $\sigma_{n}^{2}$
$p\left(f_{i} \mid f\left(x_{i}\right)\right) \propto e^{-\left(f_{i}-f\left(x_{i}\right)\right)^{2} / 2 \sigma_{n}^{2}}$

Animation by


Dr. Ádám Fekete King's College London
https://mybinder.org/v2/gh/fekad/gp-visualisation.git/master?filepath=gpvisualisation.ipynb

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Posterior Gaussian process

## GP learning: design of kernel function

- A Gaussian Process regression assumes a normal distribution for $p(\varepsilon(\rho))$, a $G P$

$$
p(\varepsilon(\rho))=\mathscr{G P}\left(0, k\left(\rho, \rho^{\prime}\right)\right)
$$

- Kernels should respects all the symmetries of the force and possess a controllable degree of prior information


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## Symmetries

- Symmetry properties should be encoded into the kernel function



## Interaction Order

- Interaction order can also be encoded into the kernel, controlling the degree of complexity

$$
\frac{\partial^{n} k_{n}\left(\rho, \rho^{\prime}\right)}{\partial \mathbf{r}_{1} \cdots \partial \mathbf{r}_{n}}=0
$$



3- and 5-body interactions

## Part II

# Choice of kernel function and computational speedups 

Claudio Zeni

Physics Department, King's College London

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3- and 5-body interactions

Building $n$-body kernels


## Building $n$-body kernels

- Permutation invariant representation of a local environment

$$
\rho\left(\mathbf{r},\left\{\mathbf{r}_{i}\right\}\right)=\sum_{i=1}^{M} \mathscr{N}\left(\mathbf{r}_{i}, \sigma^{2}\right)=
$$



- Dot product induces a 2-body kernel

$$
k_{2}\left(\rho, \rho^{\prime}\right)=\int d \mathbf{r} \rho(\mathbf{r}) \rho^{\prime}(\mathbf{r})=
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- Haar integration imposes rotational symmetry

$$
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$$

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## SOAP kernel

- Start by representing the environment of an atom as its neighbour density

$$
\rho(\mathbf{r})=\sum_{i=1}^{M} \mathcal{N}\left(\mathbf{r}_{i}, \sigma^{2}\right) f_{c u t}\left(\left|\mathbf{r}_{i}\right|\right)
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- Expand the neighbour density in a basis of spherical harmonics and radial functions:

$$
\rho(\mathbf{r})=\sum_{n l m} c_{l m n}^{i} Y_{l m}(\hat{\mathbf{r}}) g_{n}(r)
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- Take the power spectrum and normalize it:

$$
\tilde{p}_{n n^{\prime} l}=\sum_{m=-l}^{l} c_{n l m}^{i *} c_{n^{\prime} l m}^{i} \quad \mathbf{p}^{i}=\tilde{\mathbf{p}}^{\mathbf{i}} /\left|\widetilde{\mathbf{p}}^{\mathbf{i}}\right|
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$$

- The SOAP kernel is the scalar product elevated to an integer power:

$$
k\left(\mathbf{r}_{\mathbf{i}}, \mathbf{r}_{\mathbf{j}}\right)=A^{2}\left|\mathbf{p}^{i} \cdot \mathbf{p}^{j}\right|^{\xi}
$$

## Machine Learning a General-Purpose Interatomic Potential for Silicon




Machine Learning a General-Purpose Interatomic Potential for Silicon
A. P. Bartók, J. Kermode, N. Bernstein, G. Csányi (PRX 2018)

## Explicit $n$-body kernels

$$
\begin{aligned}
& k_{2}^{s}\left(\rho, \rho^{\prime}\right)=\sum_{\substack{i \in \rho \\
j \in \rho^{\prime}}} \mathrm{e}^{-\left(r_{i}-r_{j}^{\prime}\right)^{2} / 2 \ell^{2}}, \\
& k_{3}^{s}\left(\rho, \rho^{\prime}\right)=\sum_{\substack{i_{1}>i_{2} \in \rho \\
j_{1}>j_{2} \in \rho^{\prime}}} \sum_{\mathbf{P} \in \mathcal{P}} \mathrm{e}^{-\left\|\left(r_{i_{1}}, r_{i_{2}}, r_{i_{1} i_{2}}\right)^{\mathrm{T}}-\mathbf{P}\left(r_{j_{1}}^{\prime}, r_{j_{2}}^{\prime}, r_{j_{1} j_{2}}^{\prime}\right)^{\mathrm{T}}\right\|^{2} / 2 \ell^{2}} .
\end{aligned}
$$

$$
k_{M B}^{s}\left(\rho, \rho^{\prime}\right)=\mathrm{e}^{-\left(k_{3}^{s}(\rho, \rho)+k_{3}^{s}\left(\rho^{\prime}, \rho^{\prime}\right)-2 k_{3}^{s}\left(\rho, \rho^{\prime}\right)\right) / 2 \ell^{2}}
$$



- Simple low-n kernels can be easily constructed directly on invariant degrees of freedom
- Very intuitive and computationally efficient!



## Choosing the interaction order: heuristics



Amorphous Silicon: learning curves


- GP-FFs are always sensibly more accurate on target forces than traditional FFs
- Learning with a higher order kernel is more accurate but requires more data
- The optimal order depends on the material and it can be chosen as the smallest $\boldsymbol{n}$ compatible with target accuracy
- Low order models are often optimal


## Choosing the interaction order: marginal lik.

- The optimal interaction order can also be selected as the one achieving the maximal marginal likelihood

$$
\begin{aligned}
& \text { Marginal likelihood } \\
& \ln p\left(\varepsilon \mid \rho, \mathscr{M}_{n}\right) \propto-\frac{1}{2} \varepsilon^{\mathrm{T}} \mathbf{K}_{n}^{-1} \varepsilon-\frac{1}{2} \ln \left|\mathbf{K}_{n}\right| \\
& \text { Fit Complexity }
\end{aligned}
$$

Nickel systems: marginal likelihood


AG, C. Zeni, A. Fekete, A. De Vita,
Bayesian construction of n-body force fields via Gaussian process regression

## Speeding up low order models

- $n$-body kernels predictions can be decomposed in $n$-body contributions
- Hence one can tabulate the GP over the degrees of freedom of $n$ particles (e.g. distances and angles) obtaining a very fast non-parametric n-body force field
- This mapped force field (MFF) is identical to the original one but substantially faster

Mapping: Convergence of the MFF to the GP force field as a function of the number of grid points


Speedup: Prediction time of GP potential and remapped potential vs $N$ (number of training points)

C. Zeni, K. Rossi, AG, A. Fekete, N Gaston, F Baletto, A. De Vita, Building machine learning force fields for nanoclusters (JCP 2018)

## Examples of MFFs

Copper: Learnt 2-body MFF

a-Si: Learnt 3-body MFF


- Non-parametric force fields are always more accurate than parametric counterparts
- A confidence level can also be predicted, this can help to avoid extrapolation


## Automatic force field construction

I. Obtain an initial database for your system
2. Choose the order of the interaction needed (2-body, 3-body, ...)
3. Train a Gaussian Process regression with an n-body kernel
4. Tabulate and save the learned potential onto the effective degrees of freedom $\mathbf{q}$
5. Interpolate the tabulated points to yield the n-body potential energy
6. Predict the energy of a new configuration with the learned $n$-body as a classical potential

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https://github.com/kcl-tscm/mff/

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6. Predict the energy of a new configuration with the learned $n$-body as a classical potential
(7. If the GP uncertainty is too large, run an additional quantum calculation)

https://github.com/kcl-tscm/mff/

## Nanocluster MD

- GP was trained on a set of Ni 19 nanoparticles.
- 3-body MFF containing information of 1000 configurations.
- Melting of Nil9 was observed, the presence of a slush state confirmed.
- 61 million MD steps were simulated in 4 days on a 24 cores.
- With DFT it would have taken 2000 years: $10^{6}$ speed factor.




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- With DFT it would have taken 2000 years: $10^{6}$ speed factor.




## Conclusions

- For robustness and interpretability, we need to be able to control the complexity of learning algorithms
- A way to do so is to include all physical symmetries and appropriately restrict the modelled interaction order
$k_{n}^{s}\left(\rho, \rho^{\prime}\right)=\int_{O(3)} d \mathscr{R} k_{n}\left(\rho, \mathscr{R} \rho^{\prime}\right)$
- This can be done within GP regression by using fully symmetric $n$-body kernels
- Low order models are often found to be sufficiently accurate and should hence be selected for their better extrapolation properties
- Furthermore their predict ions can be mapped onto explicit bases, giving rise to fast and accurate MFFs: nonparametric force fields


## People we need to thank



Prof. Alessandro De Vita King's College London, University of Trieste


Dr. Ádám Fekete
King's College London


## Questions?

