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Predicting material properties without knowing where the atoms are.

Computational materials screening

- Finding new and better materials for
 - solar cells
 - solar to fuel conversion
 - catalysis
 - structural materials
 - electronics

Some questions:

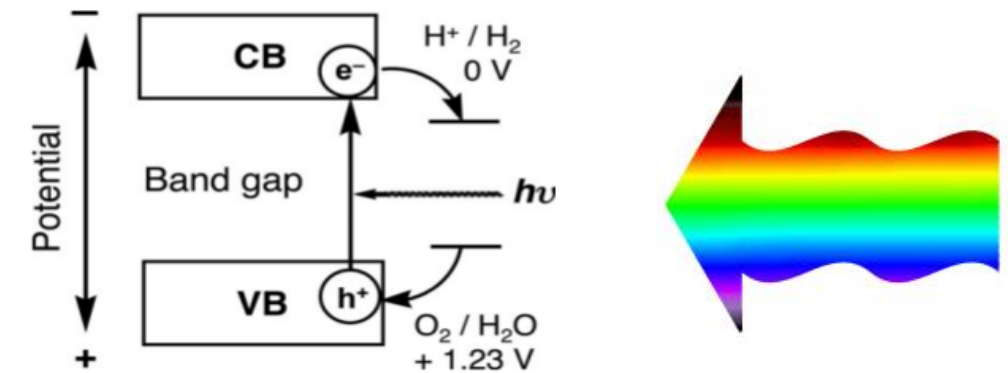
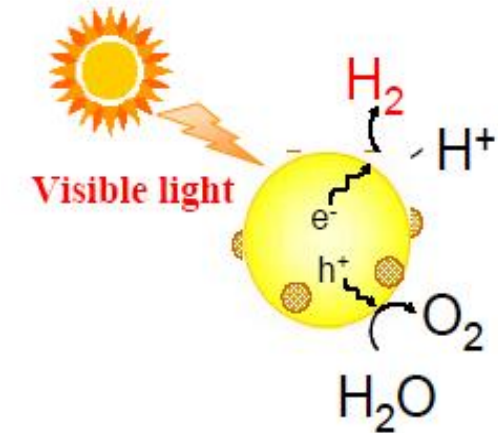
- Which materials properties should be computed?
 - “descriptors”
- How should the descriptors be computed (DFT, many-body perturbation theory...)?
- How should one search in materials space?

- Can machine learning be used to speed up computational screening?

An example: light-induced water splitting

Descriptors:

- **Stability of material**
 - Heat of formation
- **Good light absorption**
 - Bandgap in the visible range
- **Photogenerated charges at right potentials**
 - Band edges straddle the water redox potentials

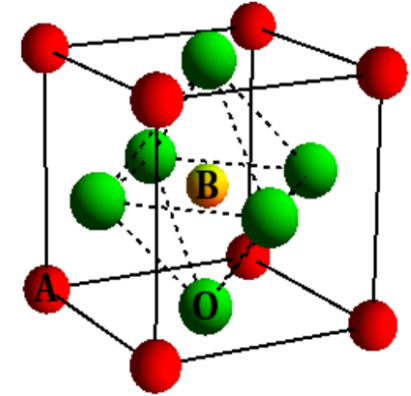


Principle of water splitting using semiconductor photocatalysts.

(Fujishima and Honda, Nature 1972)

Cubic perovskites ABX_3

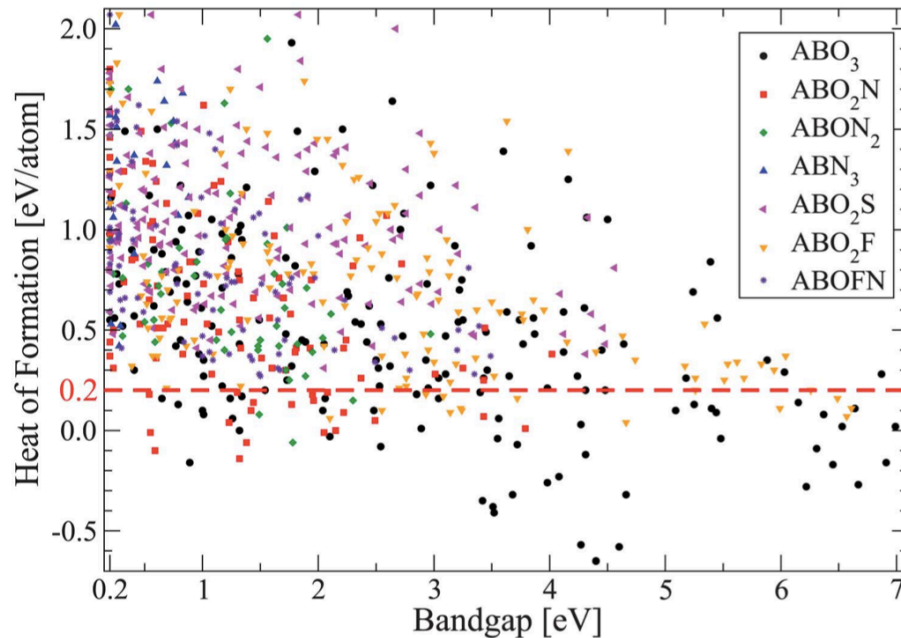
($X_3 = O_3, O_2N, ON_2, N_3, O_2S, O_2F, OFN$)



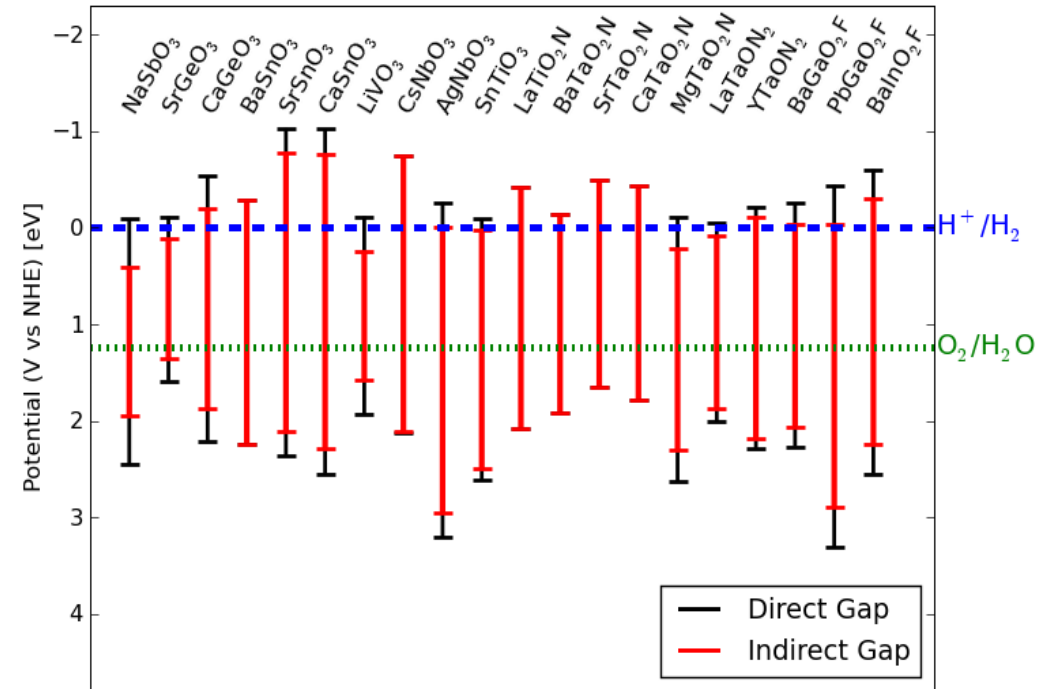
Screening criteria:

- Stability (heat of formation)
- Band gap
- Band alignment

~19000 materials



20 candidate materials
About half are known

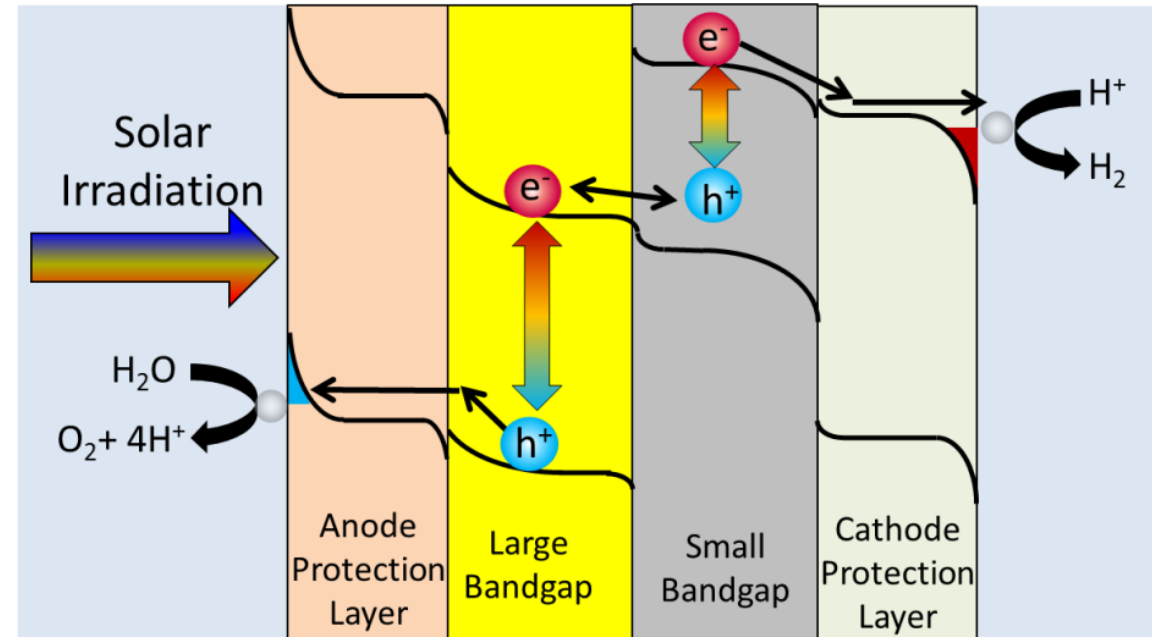


Castelli, Olsen, Datta, Landis, Dahl, Thygesen, Jacobsen, *Energy & Environmental Science*, 5(2), 5814 (2012).
Castelli, Landis, Thygesen, Dahl, Chorkendorff, Jaramillo, Jacobsen, *Energy Environ Sci* 5, 9034 (2012)

Photoelectrochemical water splitting – a more realistic device

Required materials:

- Light absorbing materials – small/large band gaps
- Protection layers
- Catalysts
- p-n junctions



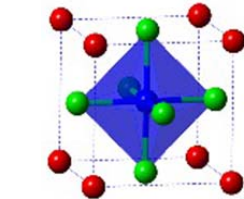
= catalyst

Small bandgap semiconductor: ~ 1.1 eV Silicon

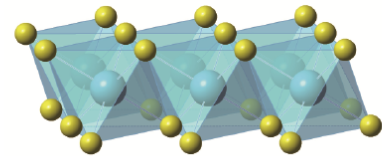
Large bandgap semiconductor: ~ 1.8 eV ??

Sulfide AB_2S_3 – screening funnel

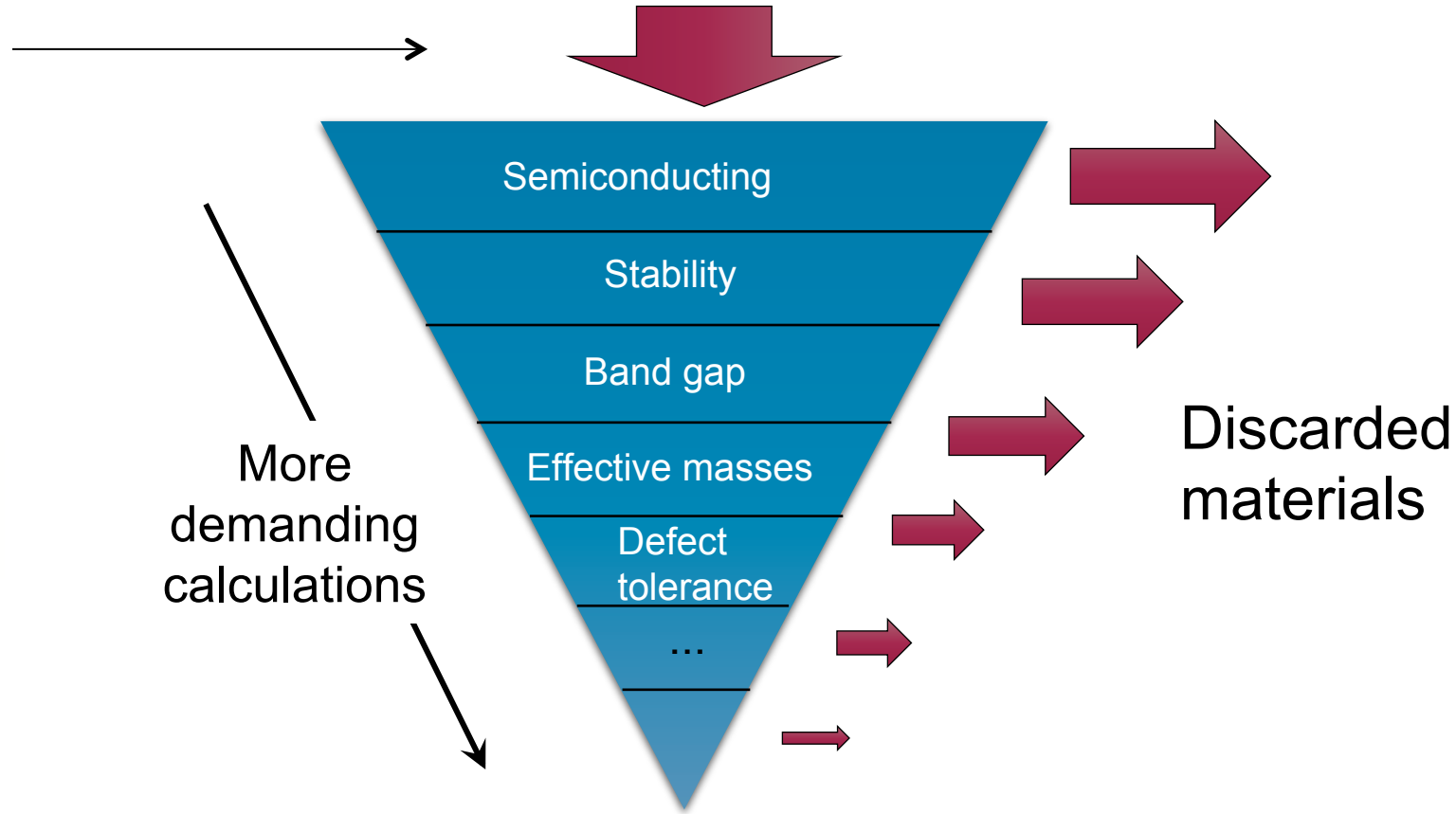
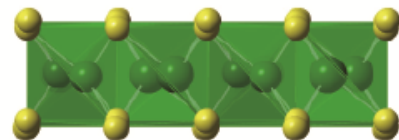
Initial structures and compositions



$53 \cdot 54 / 2 =$
1431 different
chemical
compositions



8 crystal
structures



Candidates for experimental investigation

Kuhar, Crovetto, Pandey, Thygesen, Seger, Vesborg, Hansen, Chorkendorff, Jacobsen, *Energy Environ. Sci.*, **10**, 2579 (2017)

Final list of ABS_3 sulfides for photoabsorption

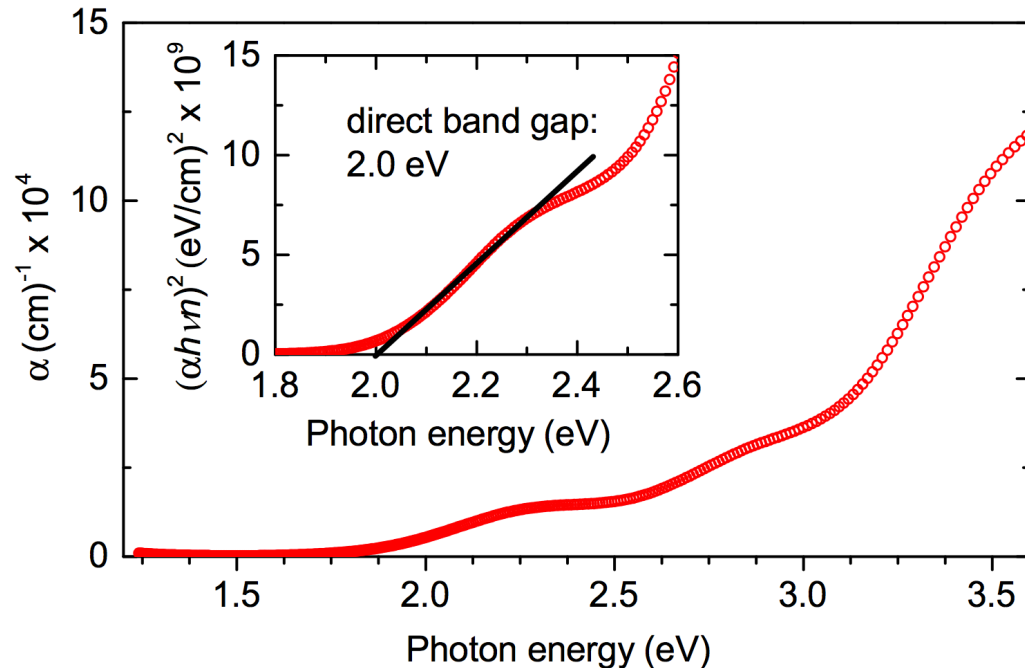
	formula	$E_g^{GLLB-SC}$	$E_{g(direct)}^{GLLB-SC}$	E_g^{HSE06}	m^*_h	m^*_e	prototype	
Initially ~ 1400 compositions x 8 structures	Ba-Hf	BaHfS ₃	1.31	1.31	1.60	-0.347	0.943	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Ba-Zr	BaZrS ₃	2.25	2.25	2.08	-0.749	0.426	GdFeO ₃
	Bi-Tl	BiTlS ₃	1.36	1.98	1.30	-0.636	0.309	FePS ₃
	Ca-Hf	CaHfS ₃	0.99	0.99	1.36	-0.336	0.759	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Ca-Sn	CaSnS ₃	1.58	1.93	1.45	-0.606	0.943	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Ca-Zr	CaZrS ₃	1.36	1.36	1.32	-0.765	0.884	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Hf-Sr	SrHfS ₃	1.12	1.12	1.45	-0.327	0.811	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Hf-Pb	HfPbS ₃	1.12	1.63	0.94	-0.275	0.235	BaNiO ₃
	La-Y	LaYS ₃	1.79	1.79	1.47	-0.670	0.490	CeTmS ₃
	Li-Ta	TaLiS ₃	1.98	2.00	2.06	-0.755	0.985	FePS ₃
Bold: all low-energy phases have relevant band gaps	Mg-Zr	MgZrS ₃	2.21	2.32	2.06	-0.718	0.779	distorted
	Sb-Y	SbYS ₃	2.03	2.09	1.67	-0.372	0.484	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Sr-Zr	SrZrS ₃	1.46	1.46	1.37	-0.644	3.115	NH ₄ CdCl ₃ /Sn ₂ S ₃
	Ta-Tl	TaTlS ₃	1.15	1.15	1.35	-0.297	0.241	distorted
	Zn-Zr	ZrZnS ₃	1.91	1.97	1.87	-0.616	0.420	FePS ₃

Kuhar, Crovetto, Pandey, Thygesen, Seger, Vesborg, Hansen, Chorkendorff, Jacobsen, Energy and Environmental Science, **10**, 2579 (2017).

BaZrS₃: W. Meng et al., Chem Mater, **28**, 821 (2016)

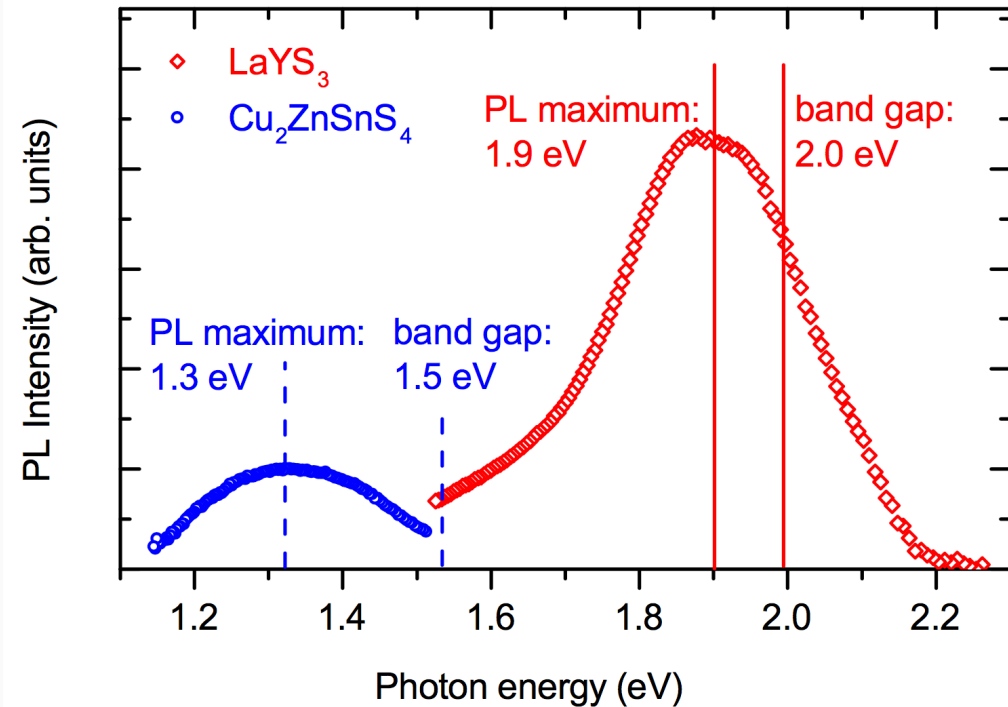
LaYS₃ - experiments

Spectroscopic ellipsometry - light absorption coefficient



Direct band gap determined from absorption coefficient and refractive index

Photoluminescence



Kuhar, Crovetto, Pandey, Thygesen, Seger, Vesborg, Hansen, Chorkendorff, Jacobsen, Energy and Environmental Science, **10**, 2579 (2017).

Screening of *known* materials for photovoltaics or water splitting

Advantages:

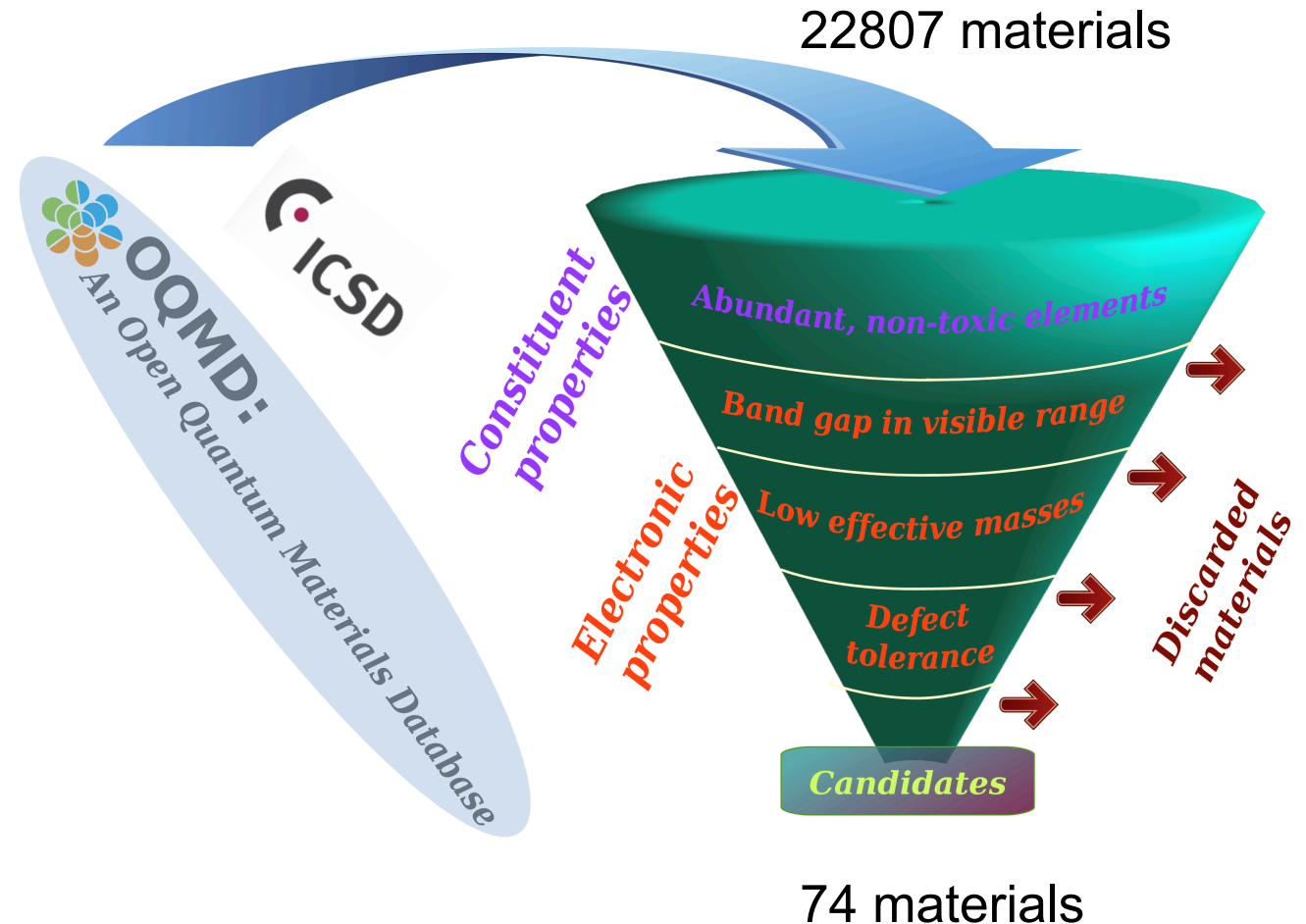
Materials known to be stable or metastable

Known synthesis procedures

Current limitations:

Binary or ternary compounds

Non-magnetic compounds



K. Kuhar, M. Pandey, K. S. Thygesen, and K. W. Jacobsen, ACS Energy Lett., 3, 436 (2018)

Screening results (74 materials)

formula	$E_g^{\text{GLLB-SC}}$ (eV)	$E_{g(\text{direct})}^{\text{GLLB-SC}}$ (eV)	m_h^* (m_e)	m_e^* (m_e)
$\text{Al}_2\text{MgSe}_4^*$	2.47	2.47	0.38	0.21
$(\text{B}_{12}\text{S})^*$	0.58	0.75	0.40	0.29
Ba_3P_4	1.07	1.07	0.95	0.97
Ba_3SbN	2.05	2.05	0.18	0.25
Antiperovskite → Ba_5Sb_4	0.94	1.27	0.66	0.36
Ba_4SnP_4	1.78	1.79	0.32	0.47
BaCaSn	0.88	0.88	0.34	0.73
BaLiP	1.98	1.98	0.16	0.16
BaZrN_2	2.45	2.45	0.38	0.28
BaZrS_3	2.34	2.34	0.35	0.43
Known perovskites → Ca_3NP	2.46	2.46	0.21	0.29
Known perovskites → CaLiSb	1.36	1.36	0.13	0.40
Known perovskites → $\text{Cs}_2\text{SnI}_6^*$	0.77	0.77	0.84	0.26
Known perovskites → Cs_3Sb	2.45	2.75	0.76	0.23
Known perovskites → Cs_6AlSb_3	2.11	2.21	0.91	0.28
Known perovskites → Cs_6GaSb_3	1.84	1.94	0.99	0.29
Known perovskites → CsCuSe_4	1.94	2.01	0.48	0.26
Known perovskites → CsGeCl_3	2.31	2.31	0.27	0.29
Known perovskites → CsNaGe_2	2.48	2.51	0.35	0.51
Known perovskites → CsSnBr_3	0.99	0.99	0.09	0.08
○	○	○	○	○
○	○	○	○	○

Database available on-line at
<https://cmr.fysik.dtu.dk>

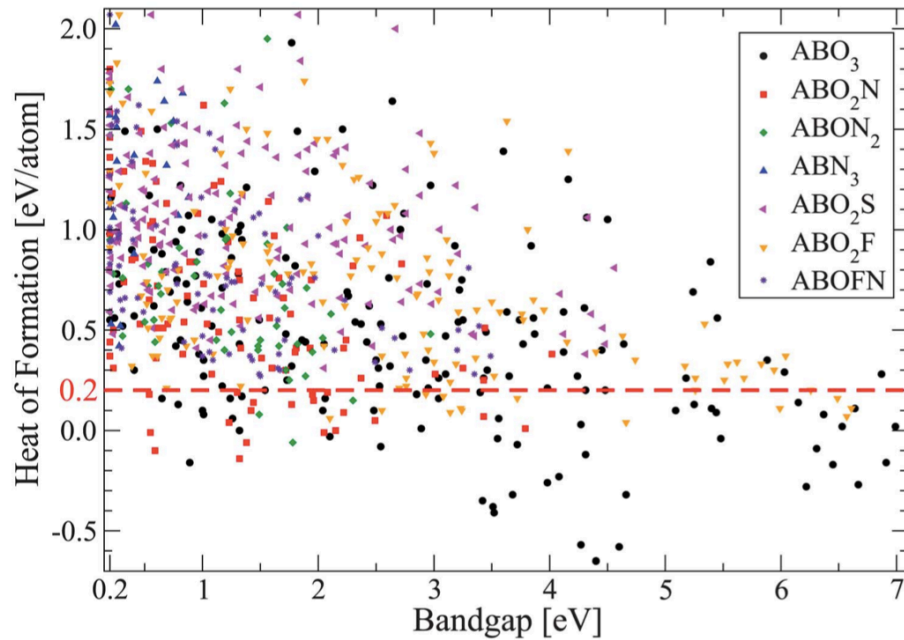
K. Kuhar, M. Pandey, K. S. Thygesen, and K. W. Jacobsen, ACS Energy Lett., 3, 436 (2018)

Back to the cubic perovskites ABX_3

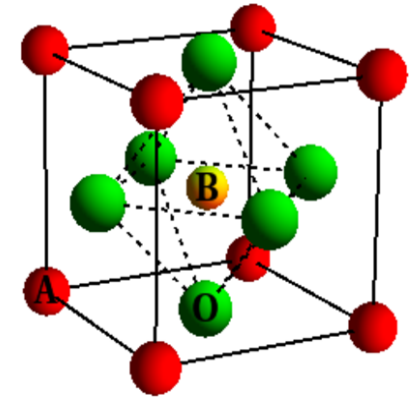
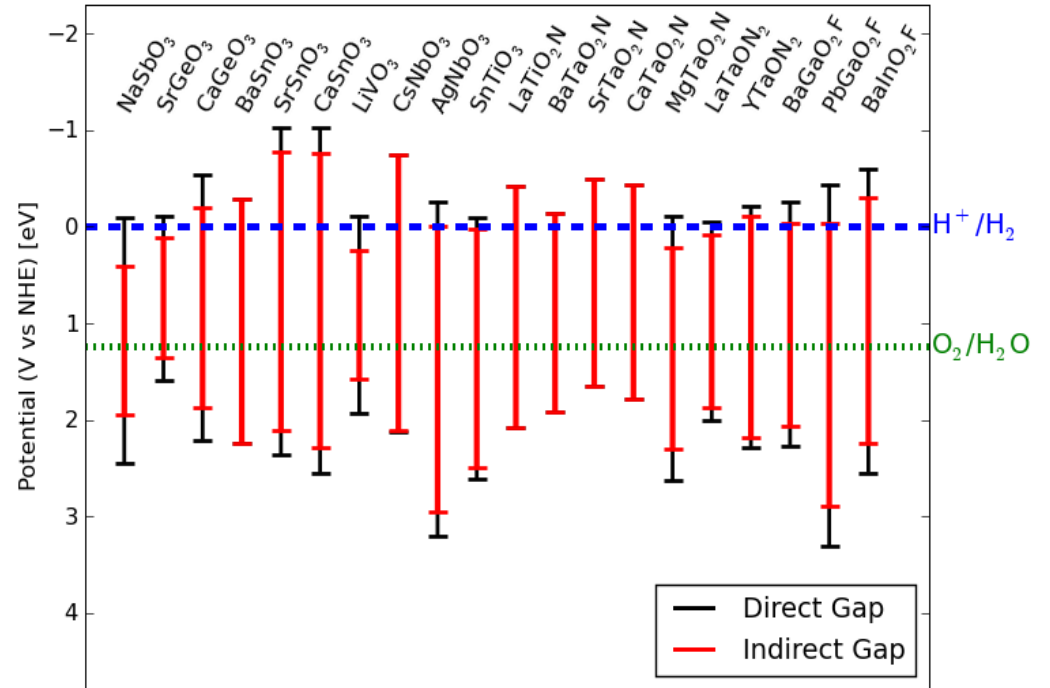
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Machine learning: Kernel regression

Fitting a function $f(x)$ based on data points $y_i = f(x_i)$

Drop a Gaussian on each data point:

$$k(x, x_i) = \exp(-|x - x_i|^2 / 2\rho^2)$$

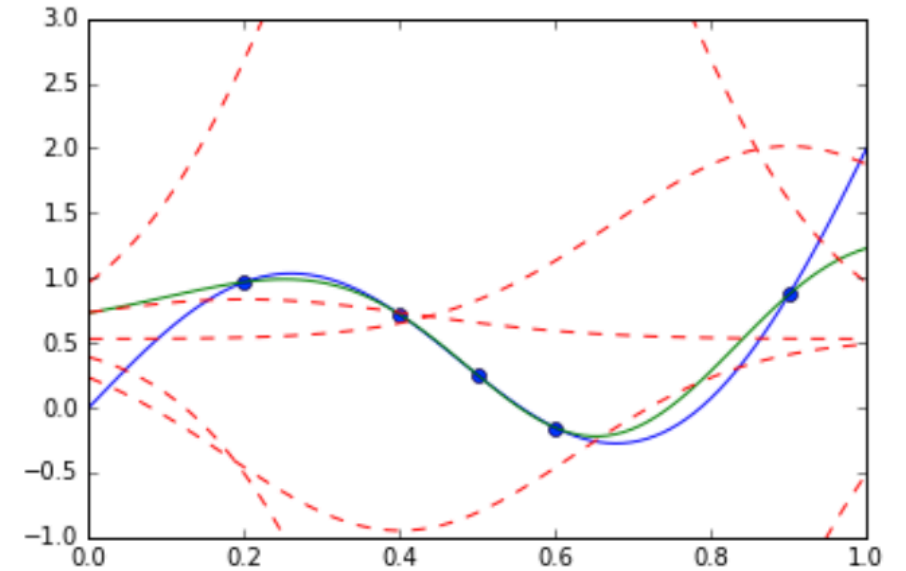
Interpolation: $y(x) = \sum_i k(x, x_i)\alpha_i$

Coefficients determined by data points:

$$y_j = \sum_i k(x_j, x_i)\alpha_i = \sum_i K_{ji}\alpha_i \rightarrow \mathbf{y} = \mathbf{K}\boldsymbol{\alpha} \rightarrow \boldsymbol{\alpha} = \mathbf{K}^{-1}\mathbf{y}$$

Interpolation: $y(x) = \mathbf{k}^T \mathbf{K}^{-1} \mathbf{y}$

with $k_i = k(x, x_i)$



Green: $f(x)$
Blue: fit
Red: Gaussians

Kernel regression with uncertainties: Gaussian process

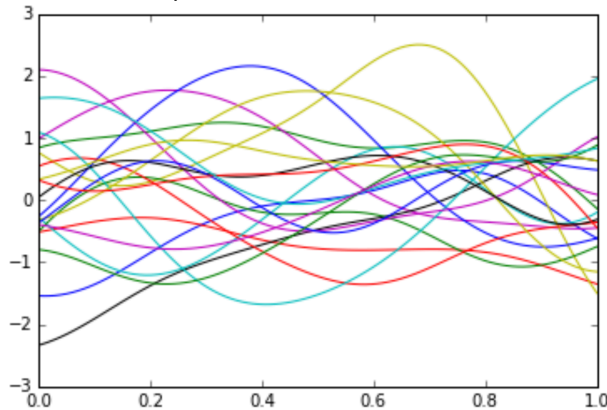
Based on Bayes theorem:
$$P(\text{Model}|\text{Data}) = \frac{1}{P(\text{Data})} P(\text{Data}|\text{Model}) P_0(\text{Model})$$

“Reinterpretation” of kernel function as correlation:
$$K_{ij} = \langle y(x_i)y(x_j) \rangle = k(x_i, x_j) = \exp(-|x_i - x_j|^2 / 2\rho^2)$$

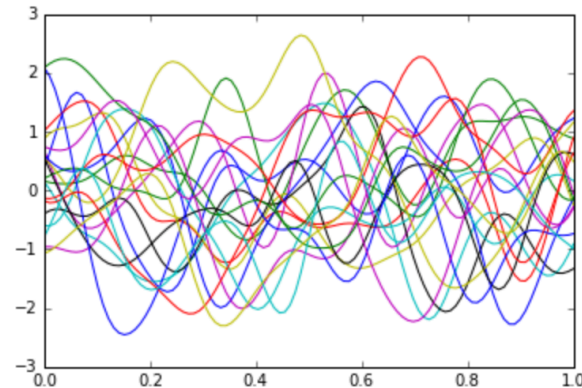
Prior probability distribution (i.e. *without* data points):

$$P_0(\mathbf{y}) = \frac{1}{\sqrt{2\pi \det(\mathbf{K})}} \exp\left(-\frac{1}{2}\mathbf{y}^T \mathbf{K}^{-1} \mathbf{y}\right), \quad \mathbf{y}^T = (y(x_1), y(x_2), \dots, y(x_N))$$

$\rho^2 = 0.1$



$\rho^2 = 0.01$



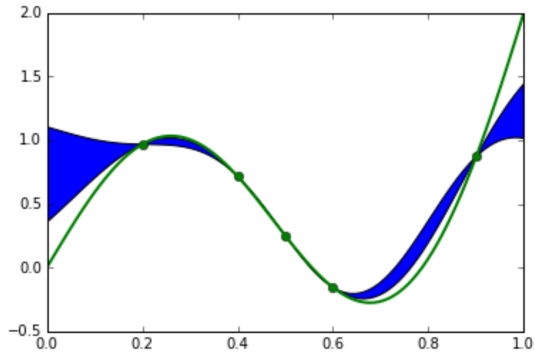
Gaussian process

Fitting a function $f(x)$ based on data points $y_i=f(x_i)$

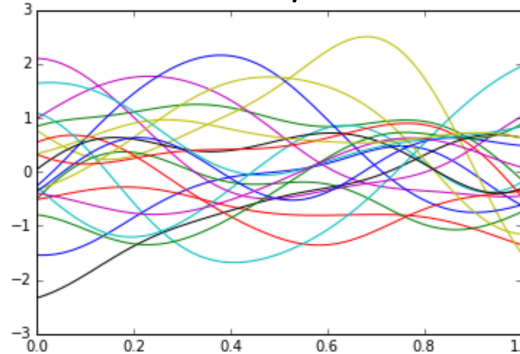
Kernel: $k(x_i, x_j) = \exp(-|x_i - x_j|^2 / 2\rho^2)$

$\rho^2 = 0.1$

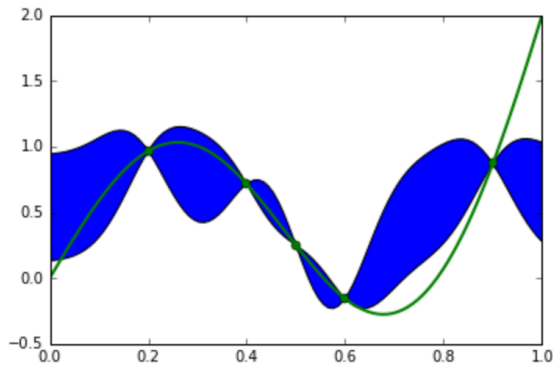
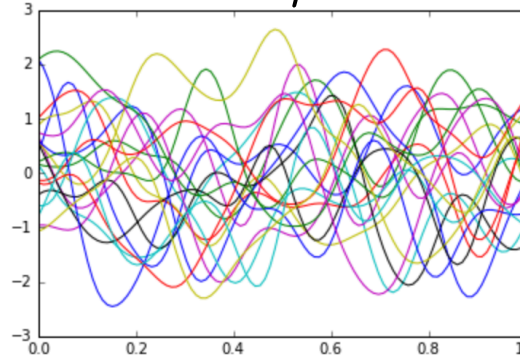
The value of ρ can be addressed by so-called cross validation



Update of model because of data

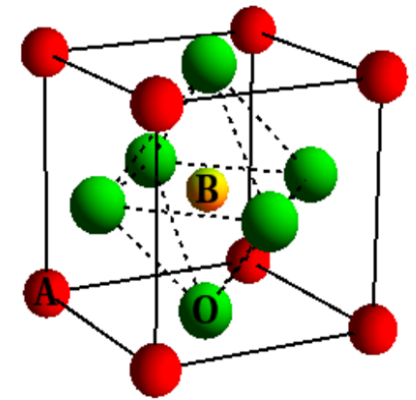
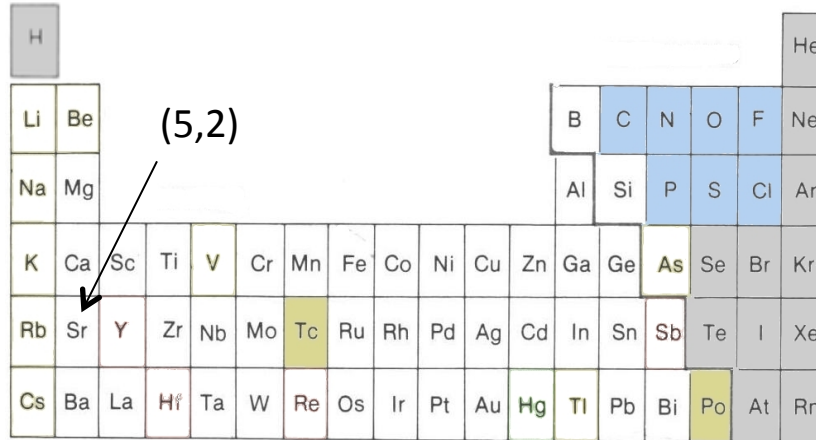
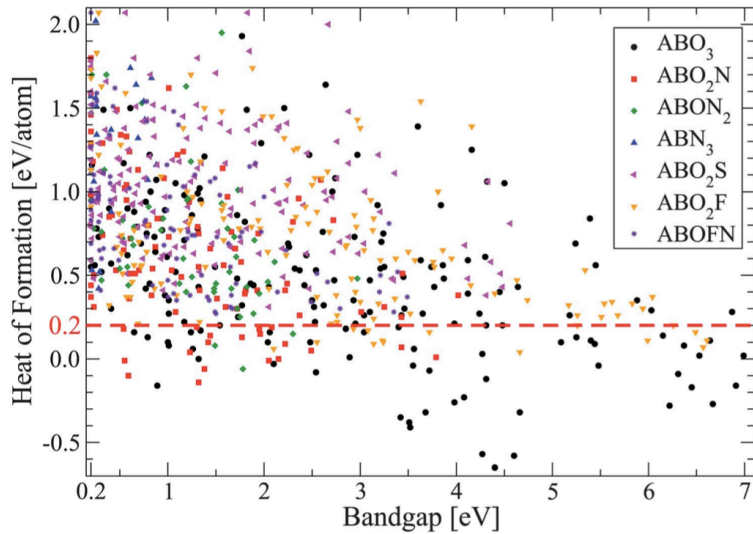


$\rho^2 = 0.01$



Back to water splitting with machine learning

About 19000 cubic perovskites oxides, oxynitrides, oxysulfides, oxyfluorides, oxyfluornitrides



$ABO_3, ABON_2, ..$

Fingerprint (x-vector):

$$x(\text{SrTaO}_2\text{N}) = (5, 2, 6, 5, 2, 1, 0, 0)$$

O, N, S, F

Sr "coordinates"

Kernel function:

$$k(x_i, x_j) = \exp(-|x_i - x_j|^2 / 2\rho^2)$$

Water splitting with Gaussian process

Training on 500 perovskites (~2.6 % of the total dataset).

Example: Heat of formation

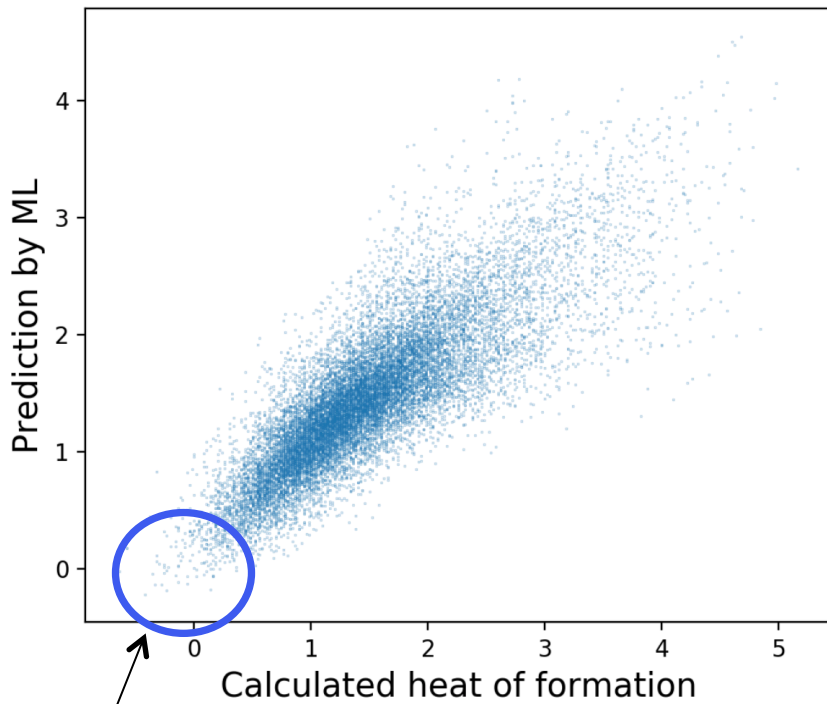
Mean Absolute Error: 0.28 eV
 Mean Absolute Predicted Error: 0.38 eV

Prediction: $y(x) = \mathbf{k}^T \mathbf{K}^{-1} \mathbf{y}$ with
 Only determined by "metric" (not by data) ← Data

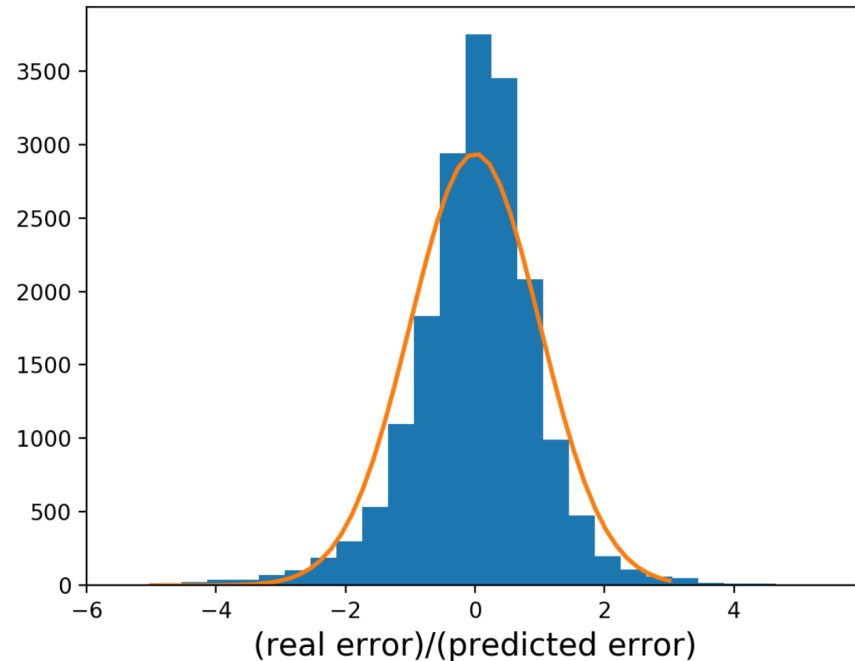
$$k_i = k(x, x_i)$$

+ error prediction

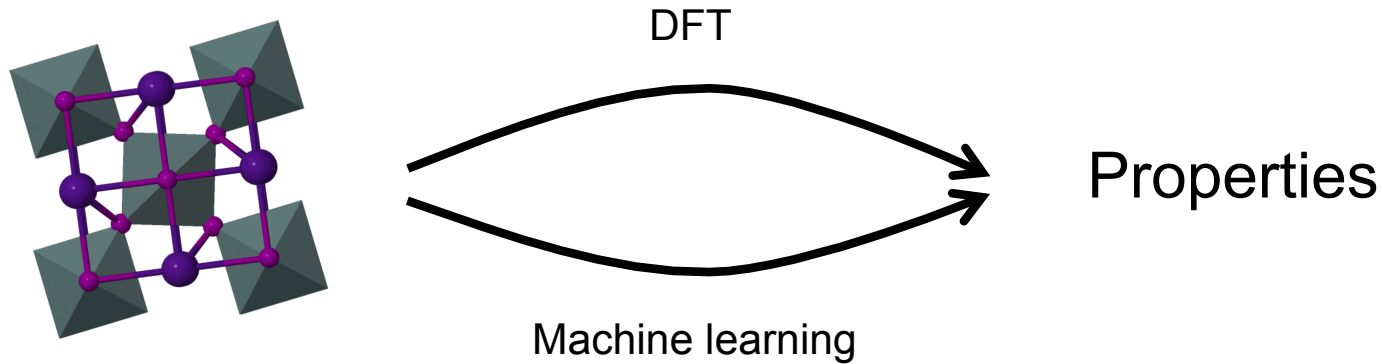
Large reduction in the number of necessary DFT calculations for stable compounds!



Stable compounds



Machine learning accelerated computational screening of *new* materials

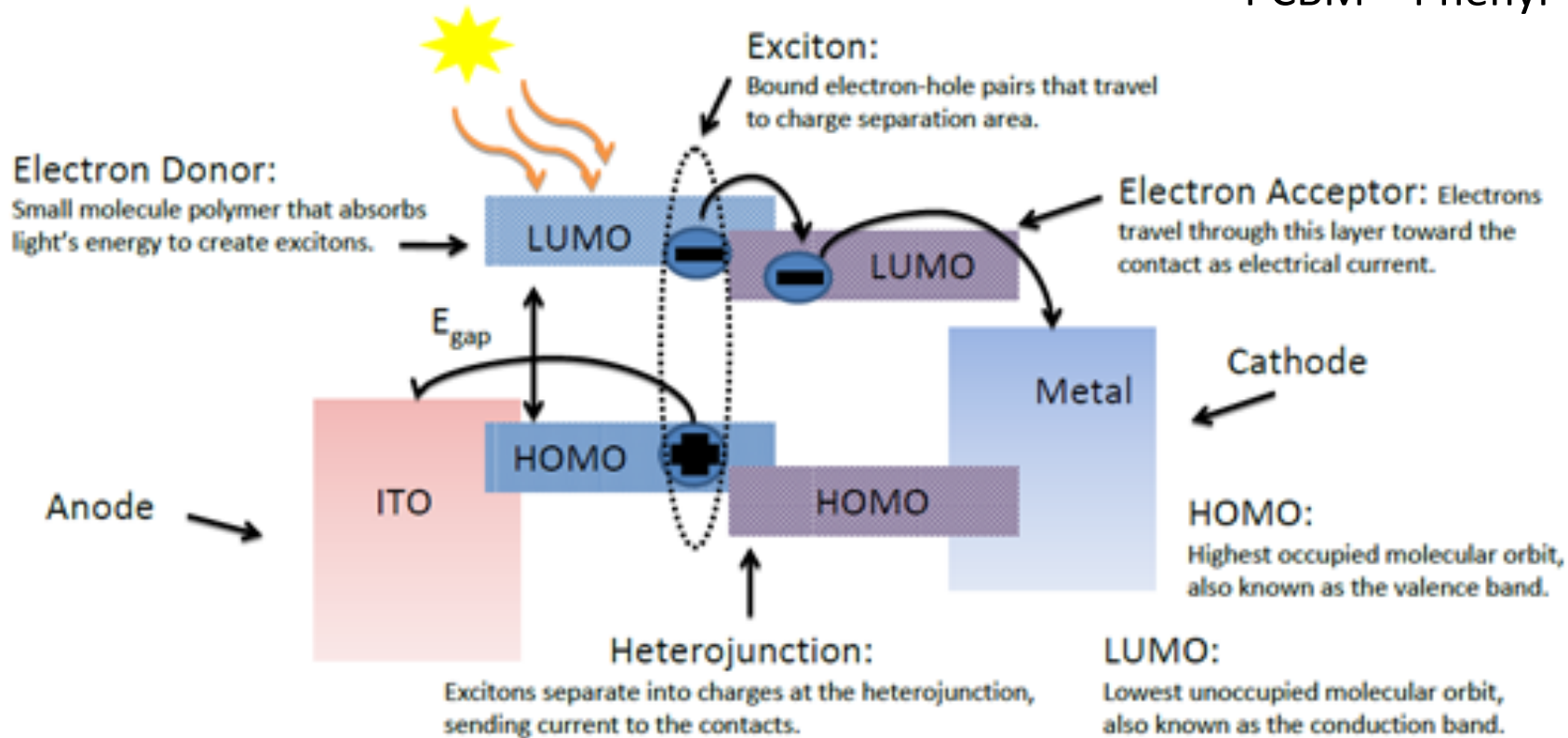


Two challenges:

- 1) Can we predict material properties for materials in many different structures where the detailed atomic positions are not known?
- 2) Can we invert the process so we go directly from properties to material? (to avoid evaluation of properties of maybe billions of (irrelevant) materials)

Organic solar cell (PCBM-based blended polymer solar cell)

PCBM = Phenyl-C₆₁-Butyric-Acid-Methyl-Ester



Peter Bjørn Jørgensen, Murat Mesta, Suranjan Shil, Juan Maria García Lastra, Karsten Wedel Jacobsen, Kristian Sommer Thygesen, and Mikkel N. Schmidt
The Journal of Chemical Physics **148**, special issue (2018)

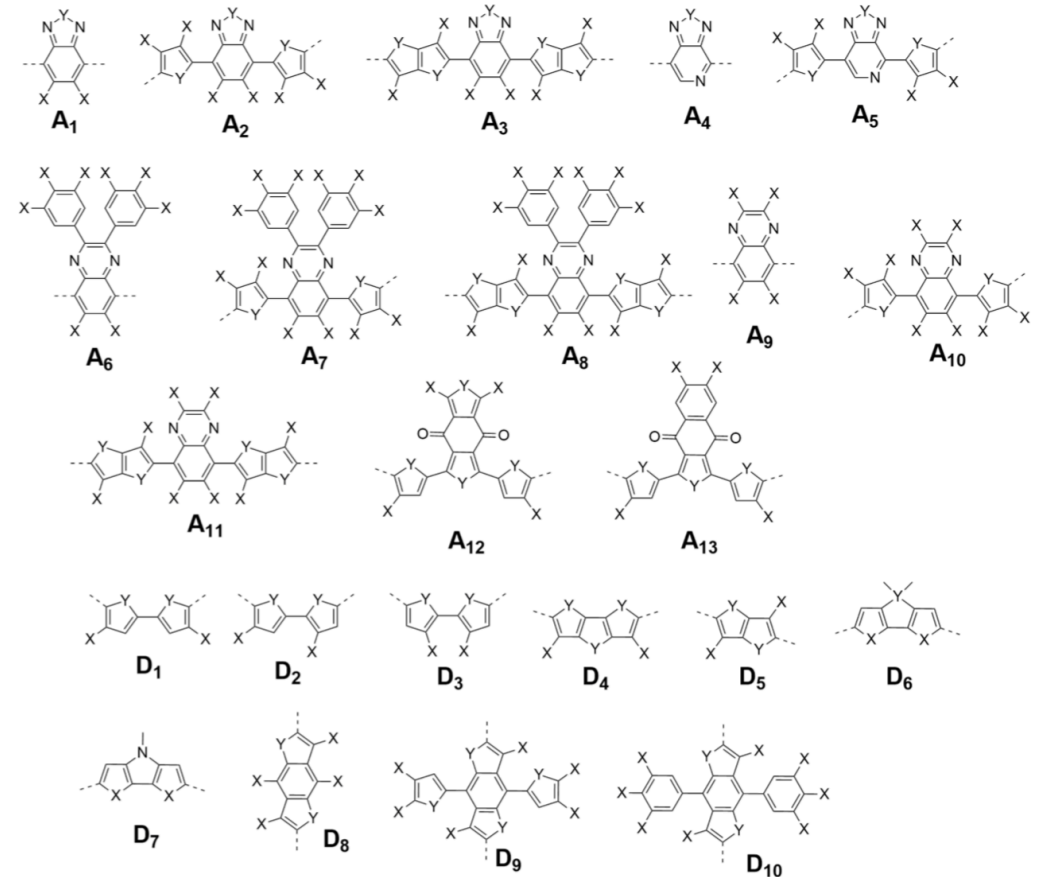
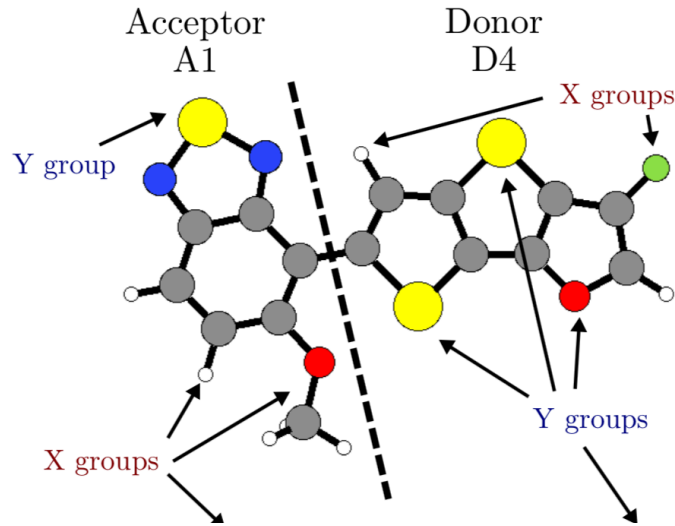
Donor-acceptor molecules (polymer units)

What is the position of the LUMO and the optical gap for these molecules?

Training set with 3989 molecules (Gaussian, B3LYP)

In principle 10^{14} molecules!

One prediction 1ms
 -> Total 10^{11} sec
 ~ 3000 years



A(1-13) = Acceptors
 D(1-10) = Donors

X = H, F, CH₃, OCH₃, SCH₃
 Y (divalent) = O, S, Se, NCH₃
 Y (tetravalent) = C, Si, Ge

Data representation

String representation of molecules.

Grammatical production rules.

No specification of atomic coordinates.

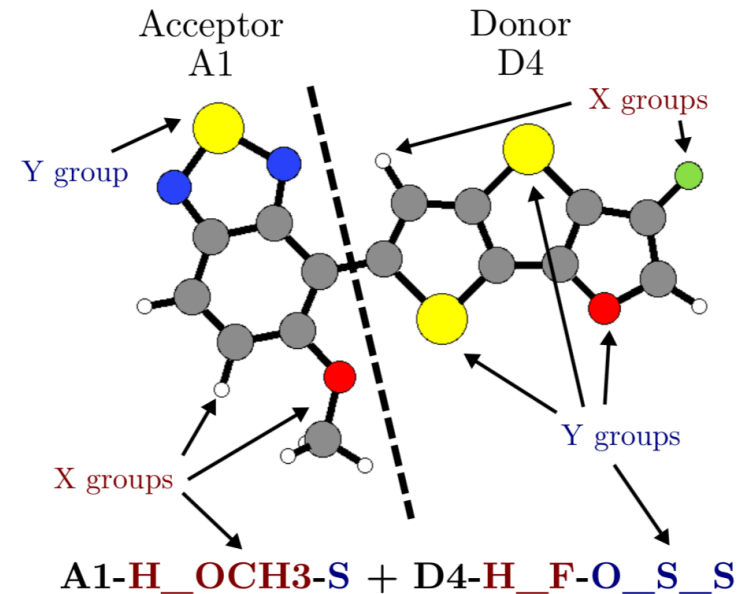


FIG. 2: String representation of one of the molecules of the solar cell dataset: “Acceptor backbone“-“X groups“-“Y groups“+“Donor backbone“-“X groups“-“Y groups“. Whenever no side groups are present “*” character is used instead.

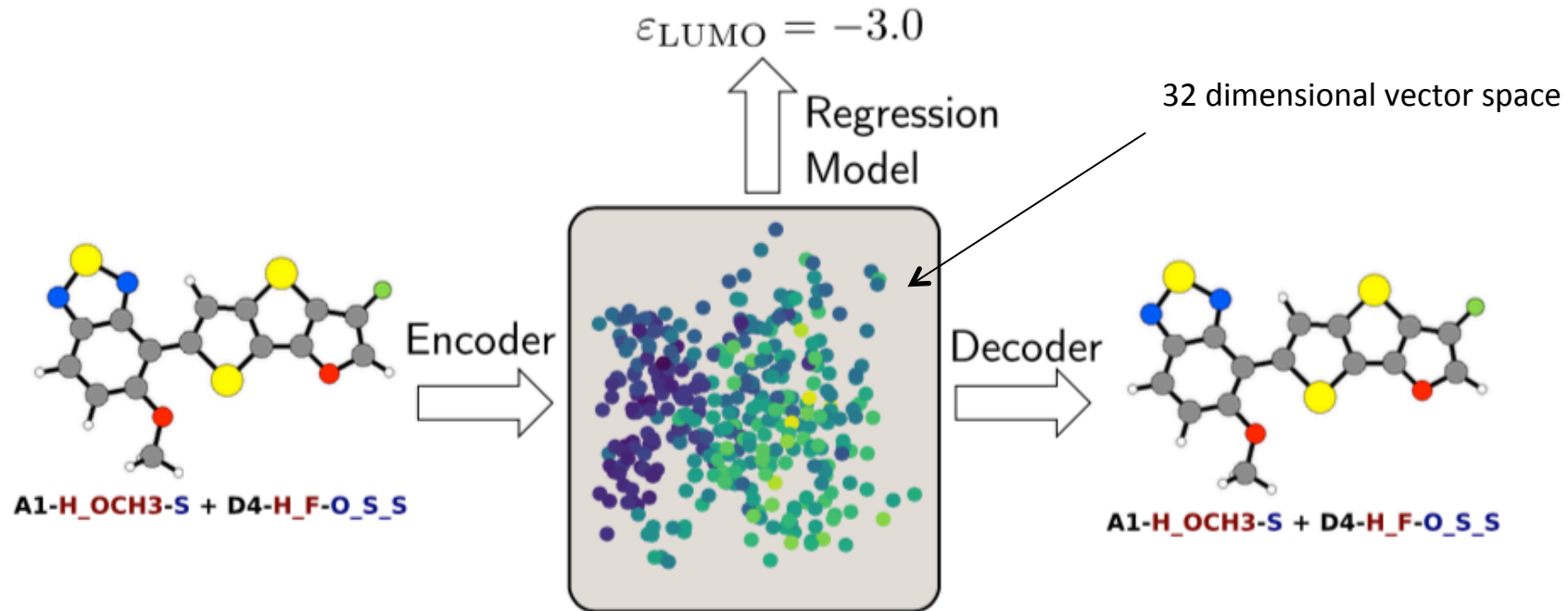
Earlier work uses SMILES to represent molecules:

Gómez-Bombarelli et al. (2016), arXiv:1610.02415 [cs.LG].

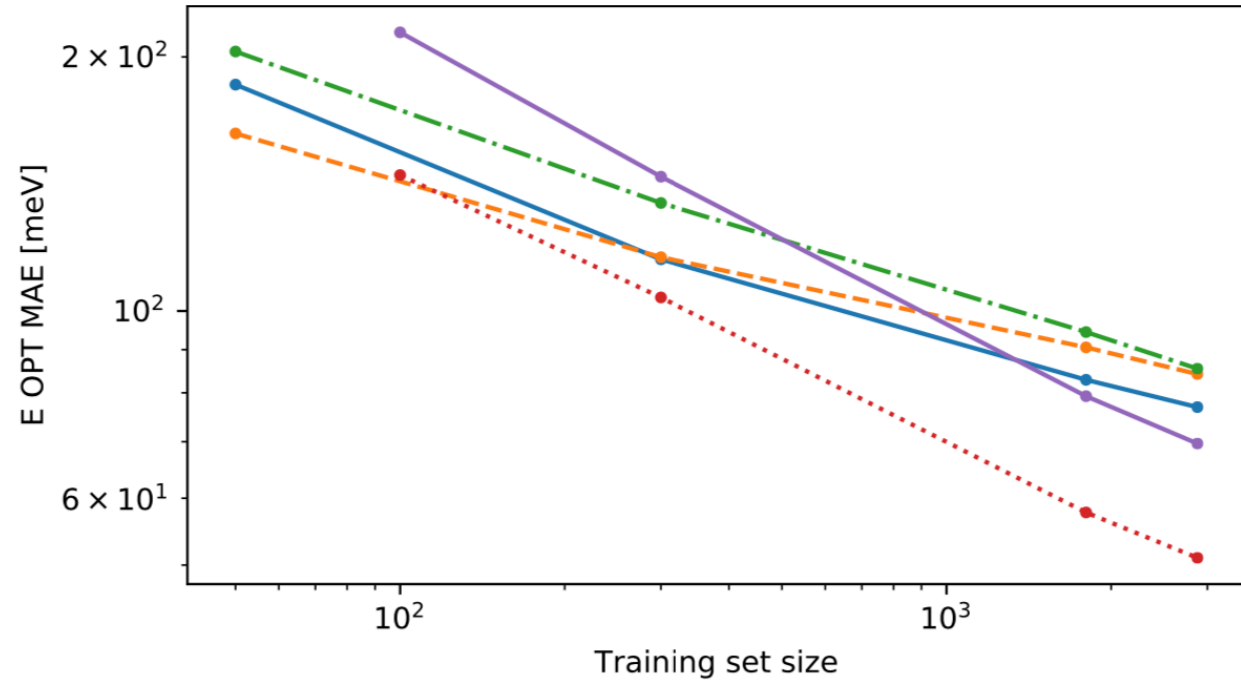
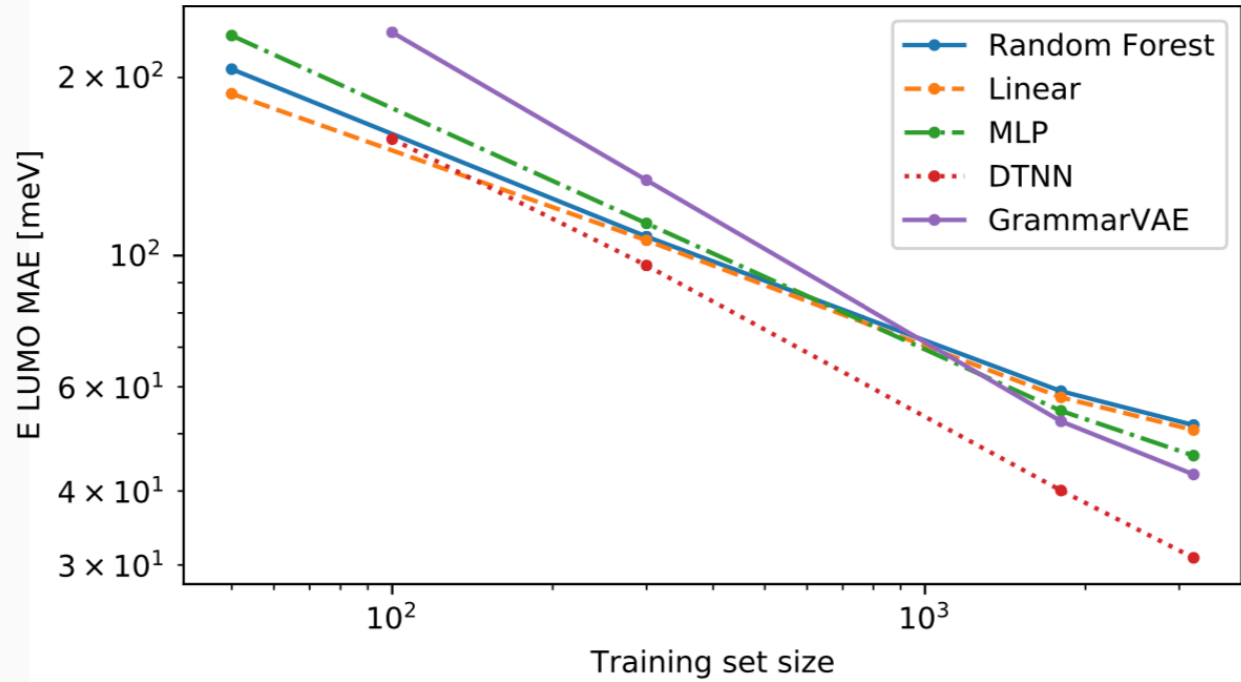
Kusner et al. (2017), arXiv:1703.01925 [stat.ML].

Variational autoencoder

Kingma and Welling [2013], Rezende et al. [2014]



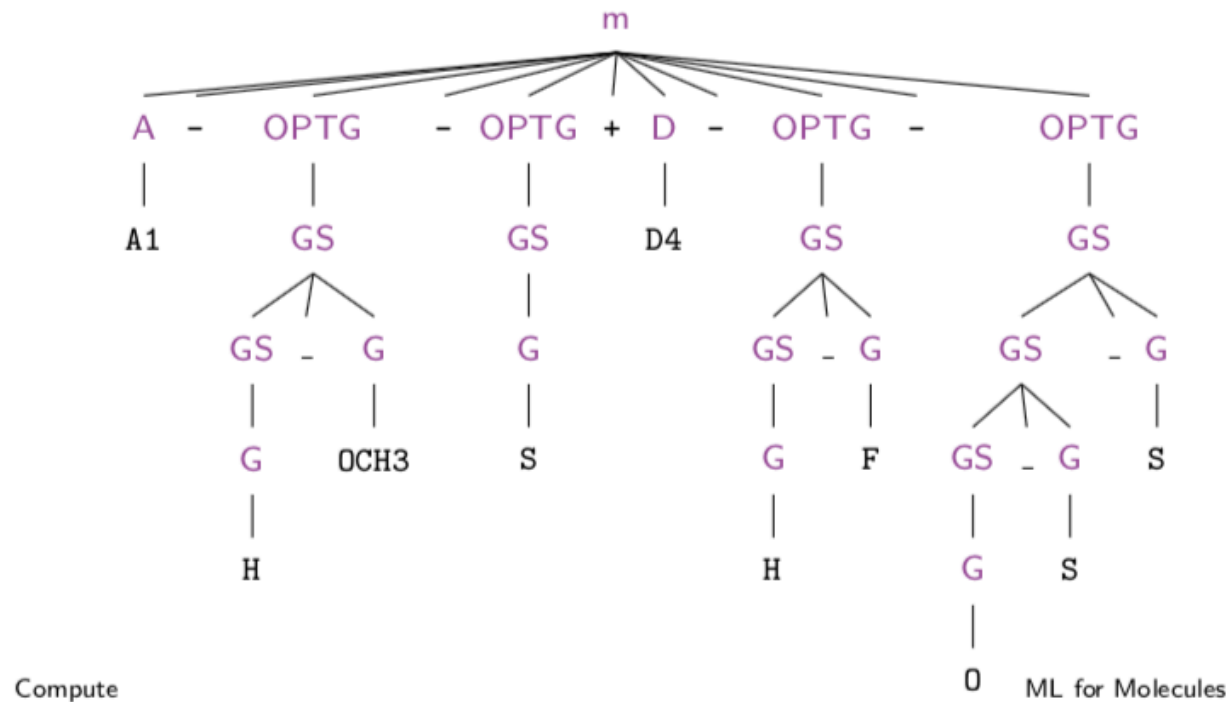
Method comparison



Grammar variational autoencoder

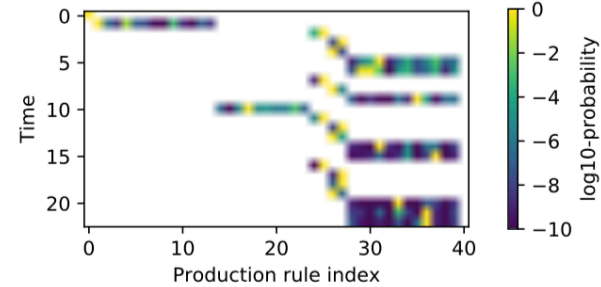
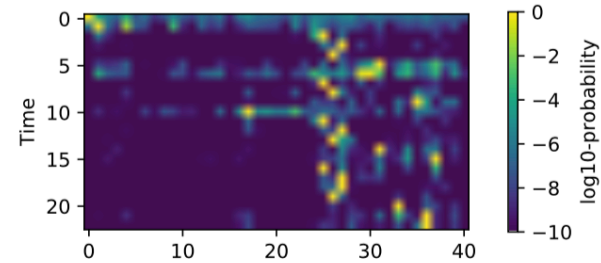
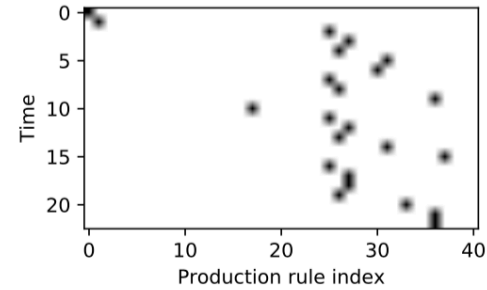
Production rules:

$m \rightarrow A - OPTG - OPTG + D - OPTG - OPTG$
 $A \rightarrow A1 | A2 | A3 | A4 | A5 | A6 | A7 | A8 | A9 | A10 | A11 | A12 | A13$
 $D \rightarrow D1 | D2 | D3 | D4 | D5 | D6 | D7 | D8 | D9 | D10$
 $OPTG \rightarrow * | GS$
 $GS \rightarrow G | GS - G$
 $G \rightarrow Ge | CH3 | OCH3 | H | C | O | SCH3 | NCH3 | S | F | Si | Se$



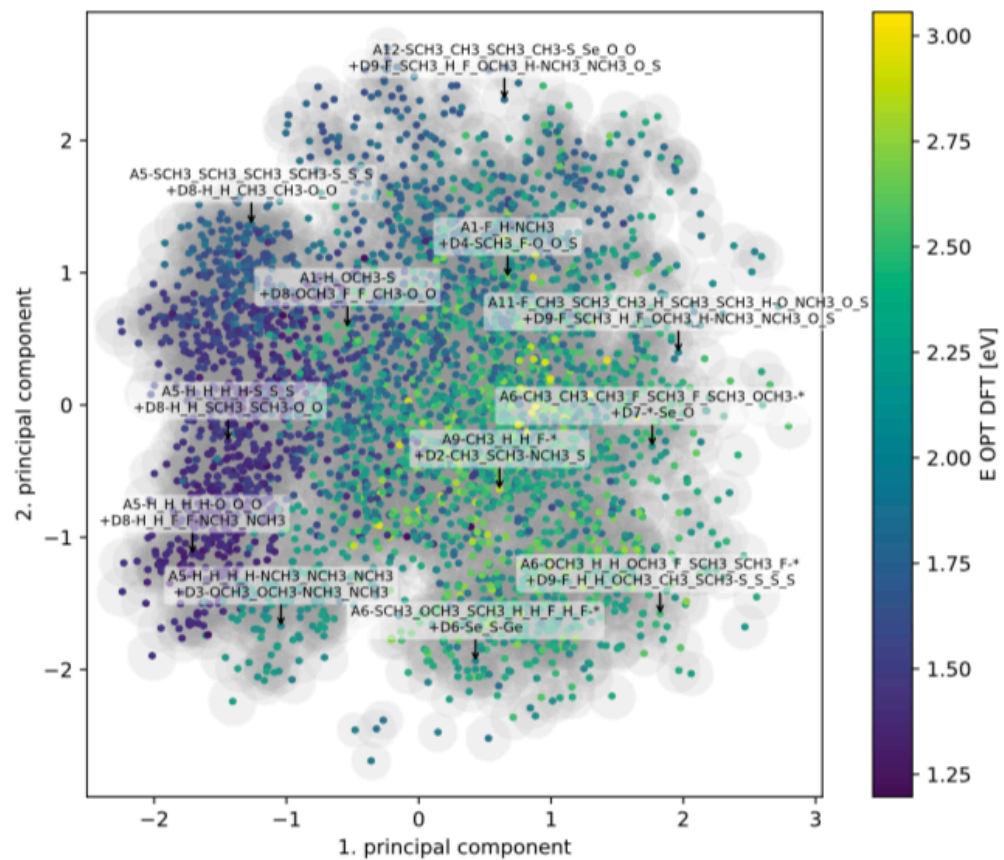
Production rule matrix encoding

m → A - OPTG - OPTG + D - OPTG - OPTG
 A → A1
 OPTG → GS
 GS → GS - G
 GS → G
 G → H
 G → OCH3
 OPTG → GS
 GS → G
 G → S
 D → D4
 OPTG → GS
 GS → GS - G
 GS → G
 G → H
 G → F
 OPTG → GS
 GS → GS - G
 GS → GS - G
 GS → G
 G → O
 G → S
 G → S

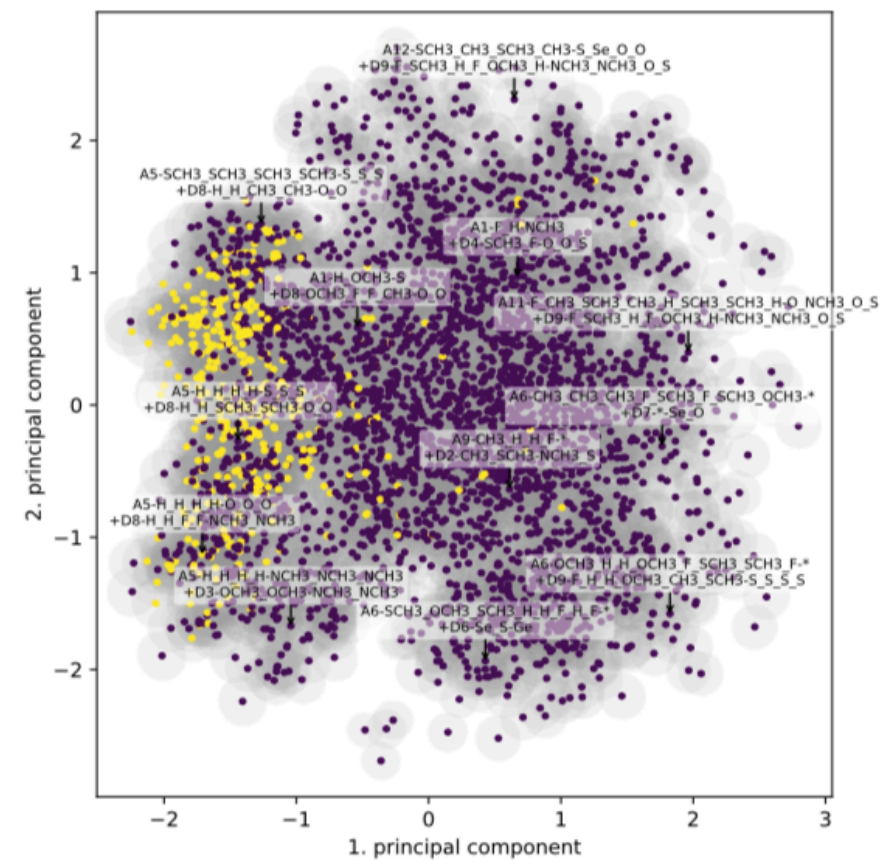


Latent space

First 2 principal components of 32-dimensional space



Colored according to optical gap

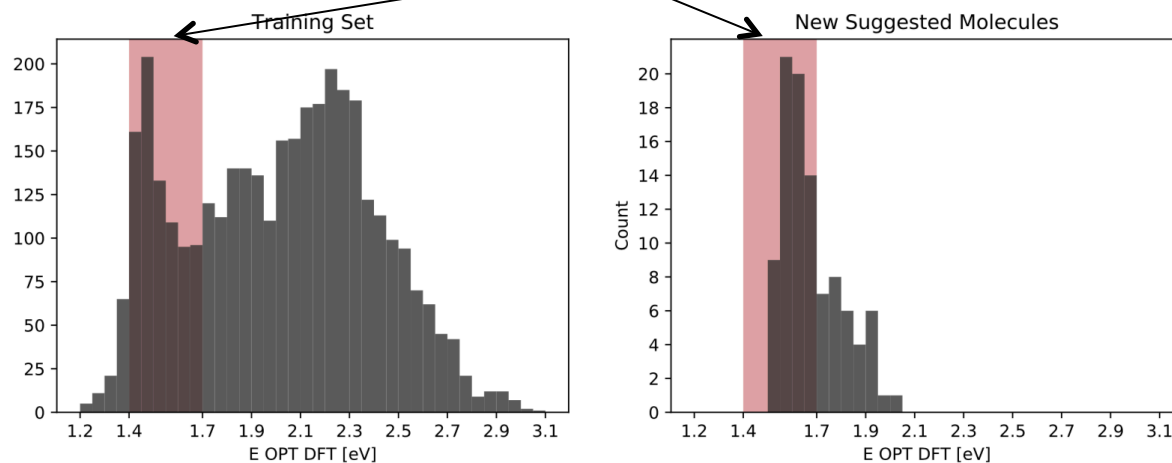


Bright points are within target range

Prediction of new molecules

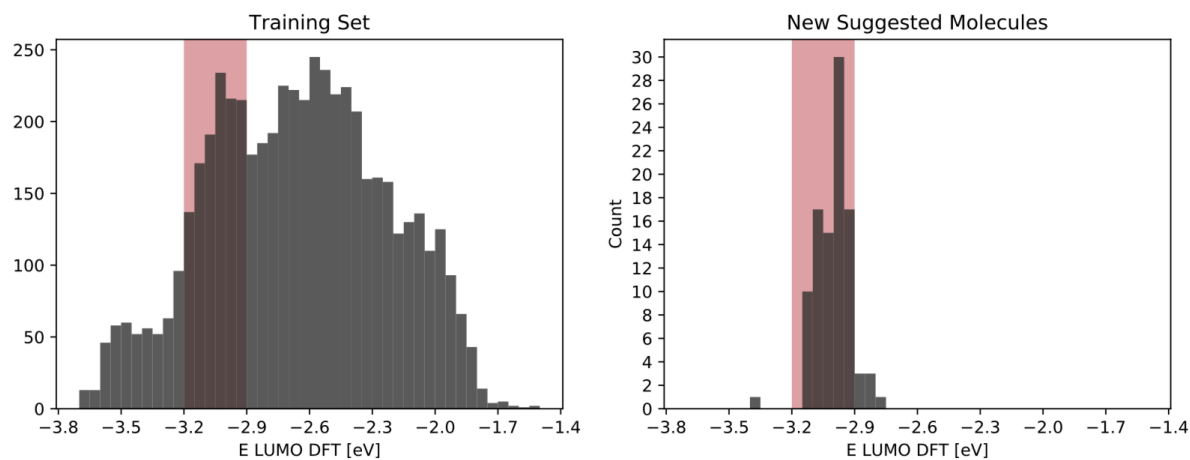
Target region

Optical band gap



100 new molecules predicted

LUMO energy

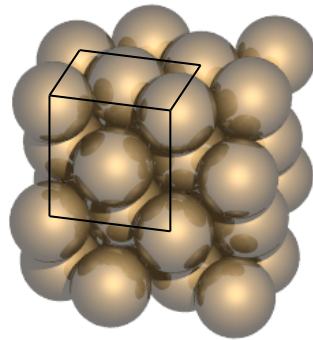


How do we classify materials without using atomic positions?

Composition, symmetry and prototypes:

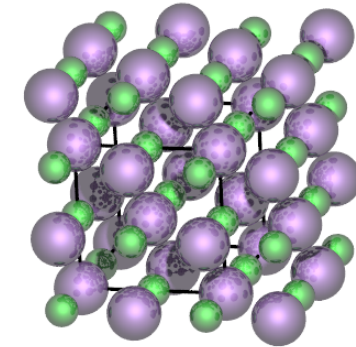
FCC Cu

Space group 225
Only variable is
lattice parameter



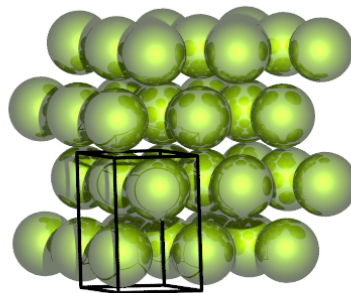
Rocksalt NaCl

Space group 225
Only variable is
lattice parameter



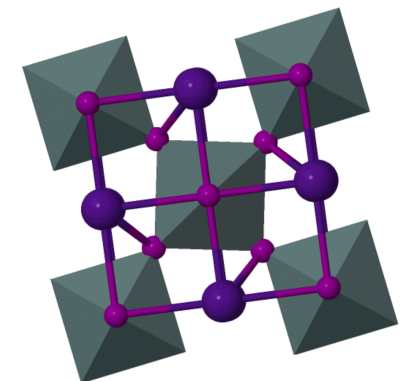
HCP Mg

Space group 194
Two lattice
parameters
 $c = 1.624 \cdot a$



CsSnI₃

Space group 127
Two lattice
parameters
Rotation angle

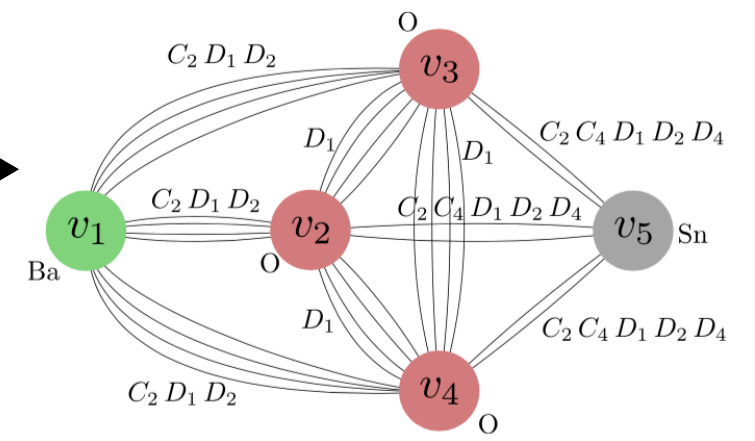
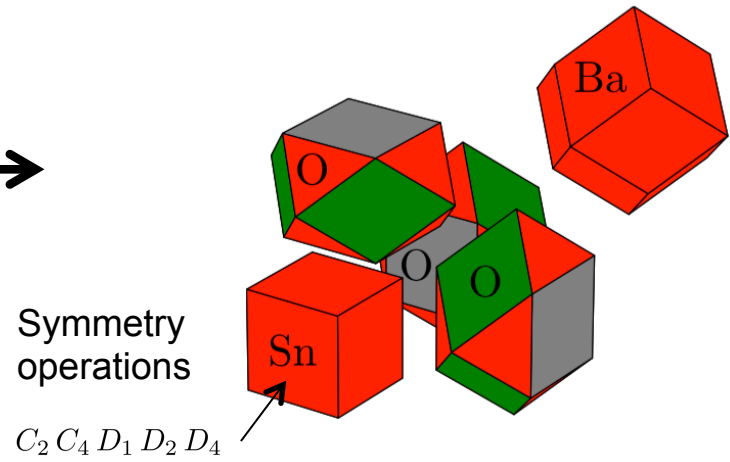
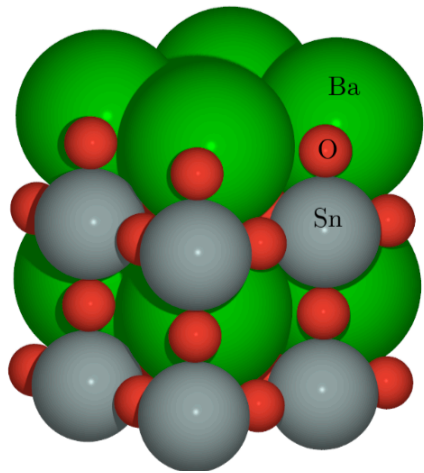


Voronoi cells and graphs

BaSnO₃ cubic perovskite

Voronoi (Wigner-Seitz) cells

Symmetry-labeled graph



Peter B. Jørgensen, Estefanía Garijo del Río, Mikkel N. Schmidt, Karsten W. Jacobsen, arXiv.org (2018)

Graph vs. prototypes

Do we know the prototype if we know the graph (and vice versa)?

Symmetry labeled graphs provide a more detailed description than prototypes

Uncertainty coefficient:
$$U(P|G) = 1 - \frac{H(P|G)}{H(P)}$$

	Graphs		Prototypes		Mutual information		Uncertainty coefficients	
	N	$ G $	$ P $	$H(G)$	$H(P)$	$I(G, P)$	$U(G P)$	$U(P G)$
Unary	1487	316	67	6.6	4.7	4.4	0.67	0.94
Binary	53528	2491	871	5.6	4.5	4.3	0.77	0.96
Ternary	339960	6927	1754	2.1	1.9	1.8	0.86	0.99

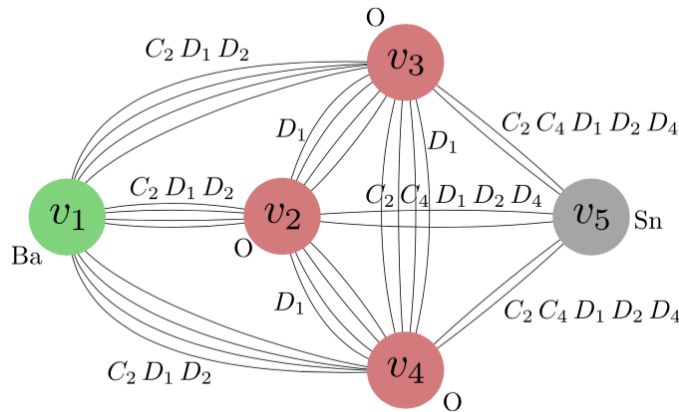
Materials from OQMD database

Message passing neural network

Only input:

Atomic numbers Z

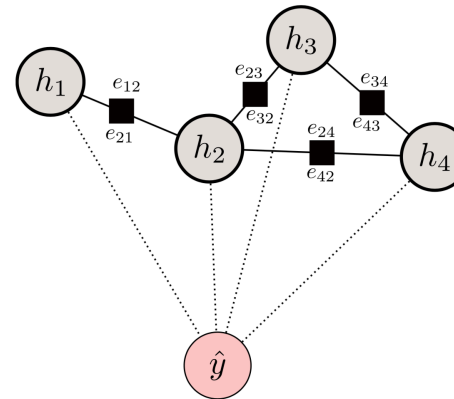
Symmetry-labeled quotient graph



atom \rightarrow node
bond \rightarrow edge



Message passing neural network



$$m_v^{t+1} = \sum_{w \in N(v)} M_t(h_v^t, h_w^t, e_{vw}^t)$$

$$h_v^{t+1} = S_t(h_v^t, m_v^{t+1})$$

$$e_{vw}^{t+1} = E_t(h_v^{t+1}, h_w^{t+1}, e_{vw}^t)$$

$$\hat{y} = R(\{h_v^T \in G\})$$

Neural networks for materials

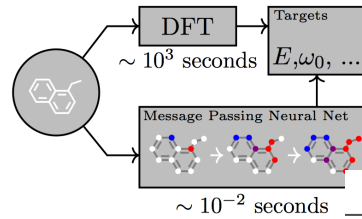
arXiv:1704.01212

Neural Message Passing for Quantum Chemistry

Justin Gilmer¹ Samuel S. Schoenholz¹ Patrick F. Riley² Oriol Vinyals³ George E. Dahl¹

Abstract

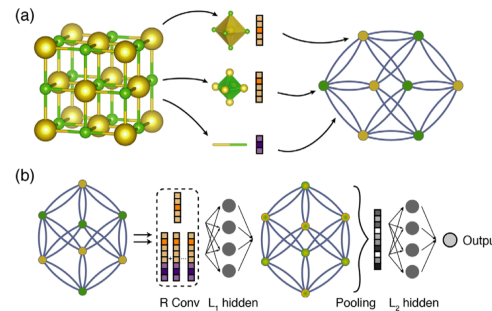
Supervised learning on molecules has incredible potential to be useful in chemistry, drug discovery, and materials science. Luckily, several promising and closely related neural network models invariant to molecular symmetries have already been described in the literature. These models learn a message passing algorithm and aggregation procedure to compute a function of their entire input graph. At this point, the next step is to find a particularly effective variant of



PHYSICAL REVIEW LETTERS 120, 145301 (2018)

Crystal Graph Convolutional Neural Networks for an Accurate and Interpretable Prediction of Material Properties

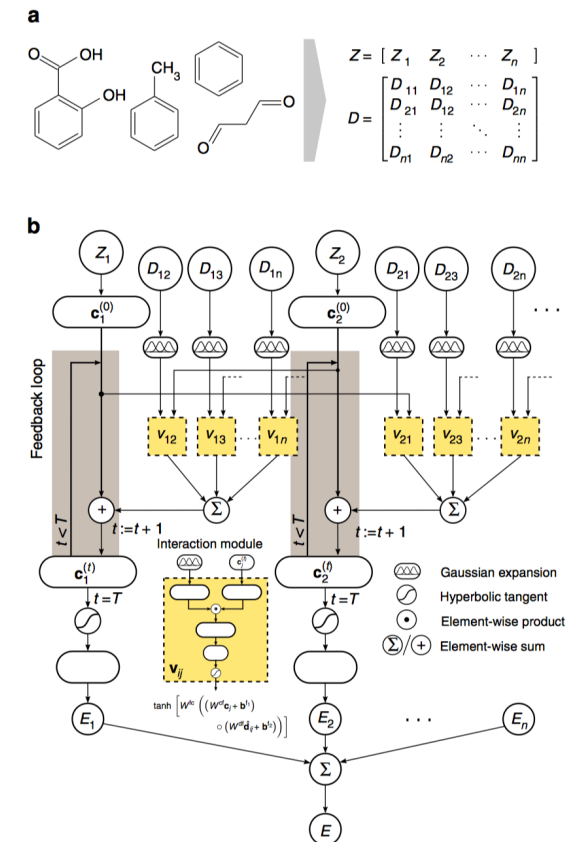
Tian Xie and Jeffrey C. Grossman
Department of Materials Science and Engineering, Massachusetts Institute of Technology,
Cambridge, Massachusetts 02139, USA



NATURE COMMUNICATIONS | DOI: 10.1038/ncomms13890

Quantum-chemical insights from deep tensor neural networks

Kristof T. Schütt¹, Farhad Arbabzadah¹, Stefan Chmiela¹, Klaus R. Müller^{1,2} & Alexandre Tkatchenko^{3,4}



Predictions on OQMD (5-fold cross validation)

~500000 DFT calculations for inorganic materials

Heat of formation:
Mean absolute error

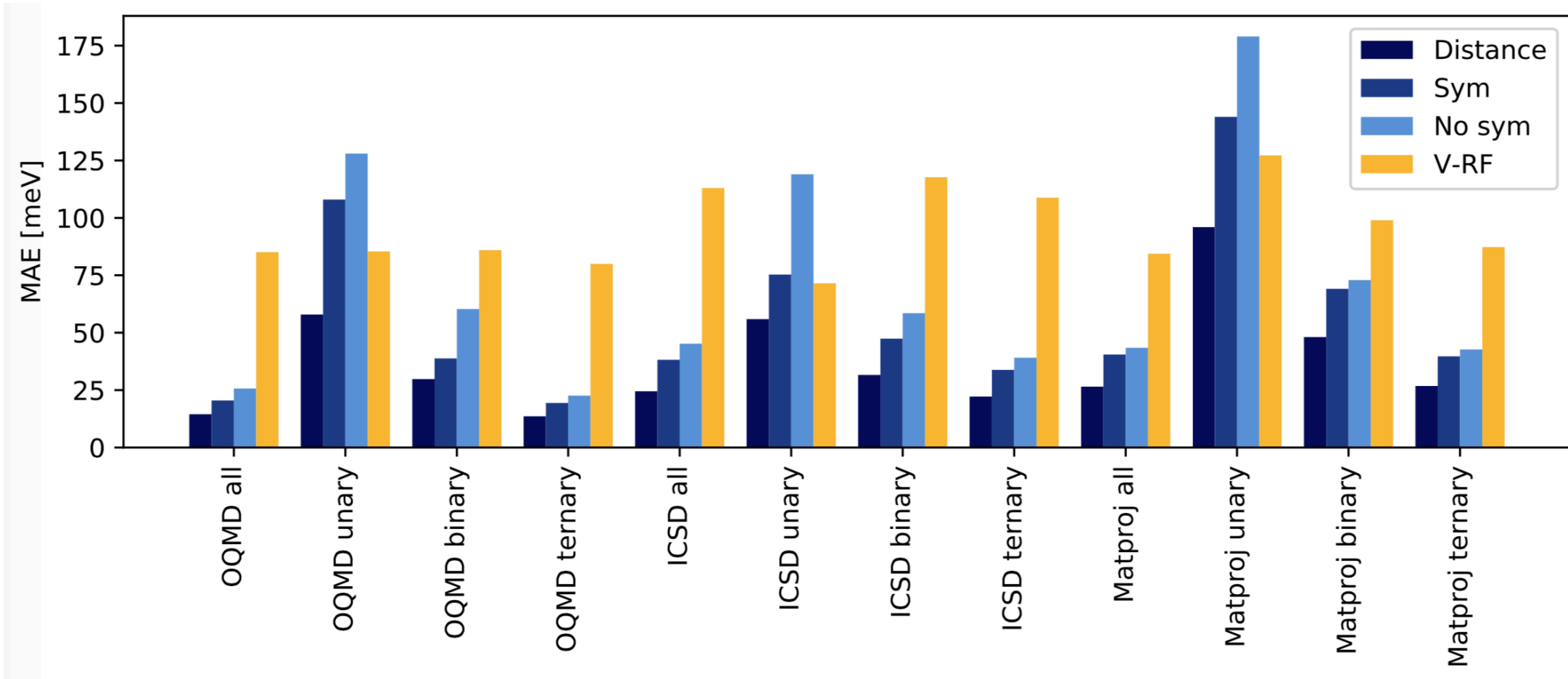
	Dist.	Sym	No sym	V-RF
OQMD all	14	20	26	85

meV!

Only Voronoi graph (+symmetry)

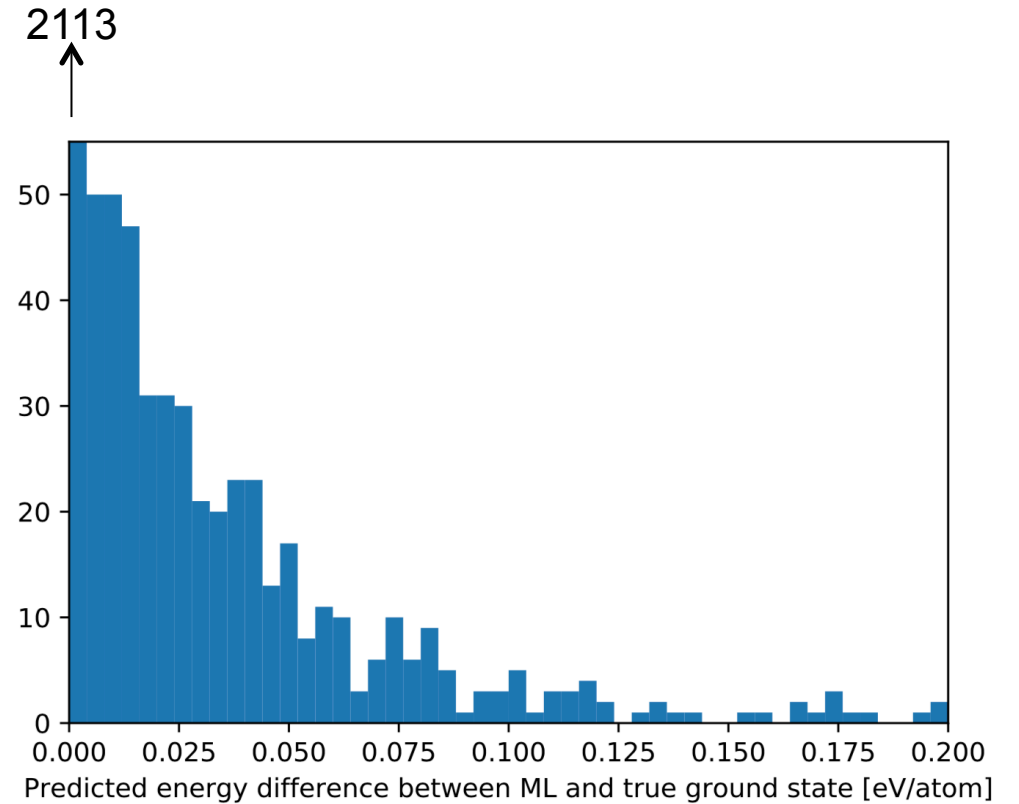
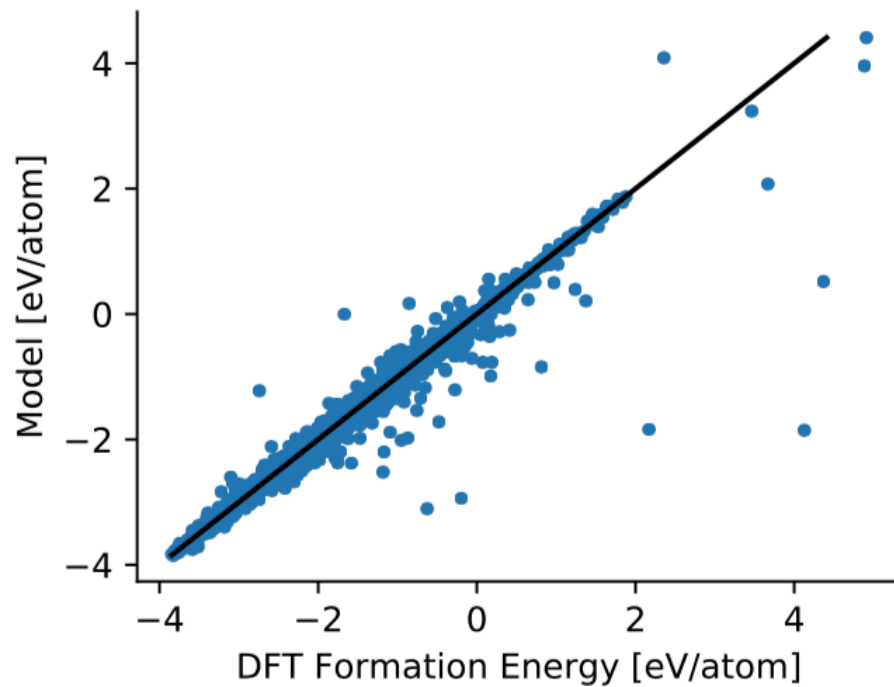
Mean absolute error on formation
energies 20 meV!
Accuracy of DFT ~100-200 meV.

Performance on subsets



The ABO_3 subset of materials

Predictions on ABO_3 oxides
MAE = 35 meV.

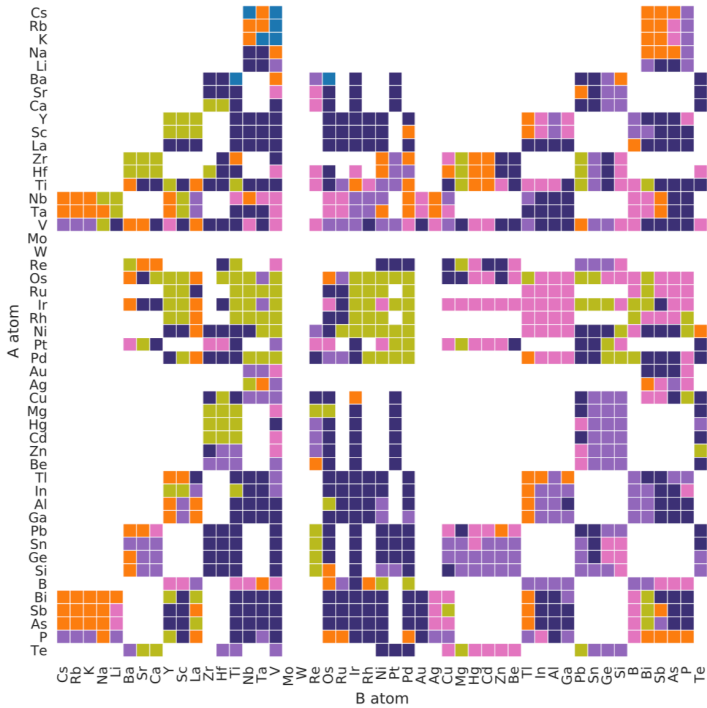


$$\Delta E = E^{ML}(G_{DFT}) - E^{ML}(G_{ML})$$

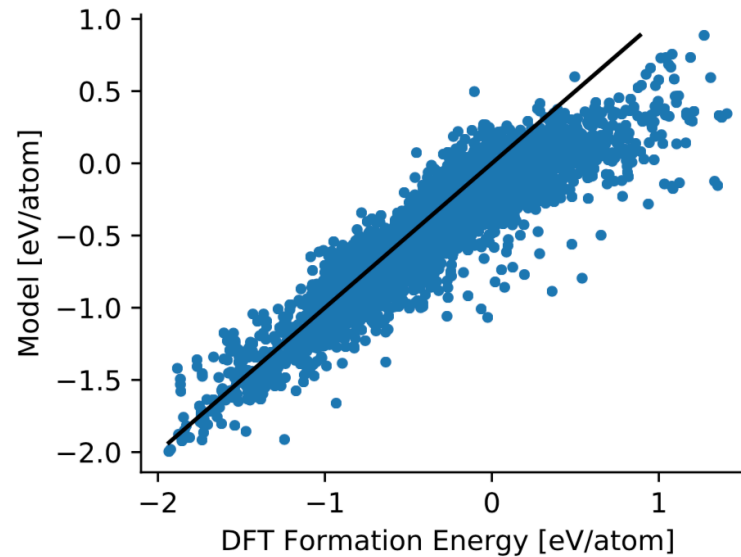
Heat of formation for ABSe₃ dataset

Only 6 ABSe₃ selenides in OQMD

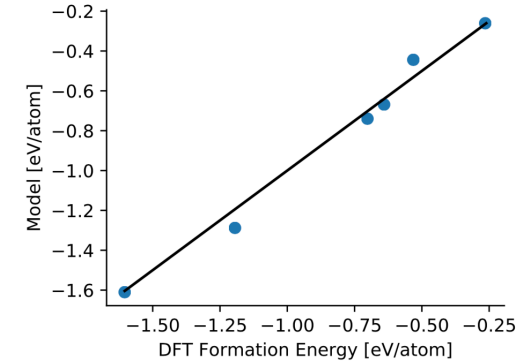
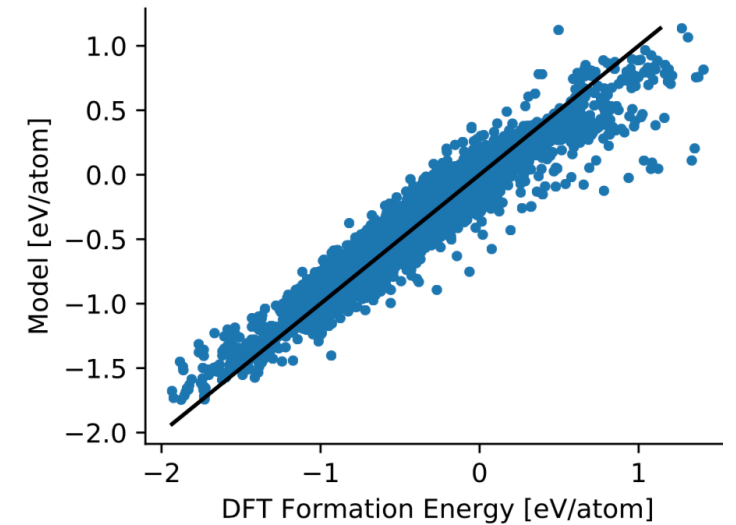
1000 compositions x 6 structures



Only training on OQMD
MAE = 0.17 eV

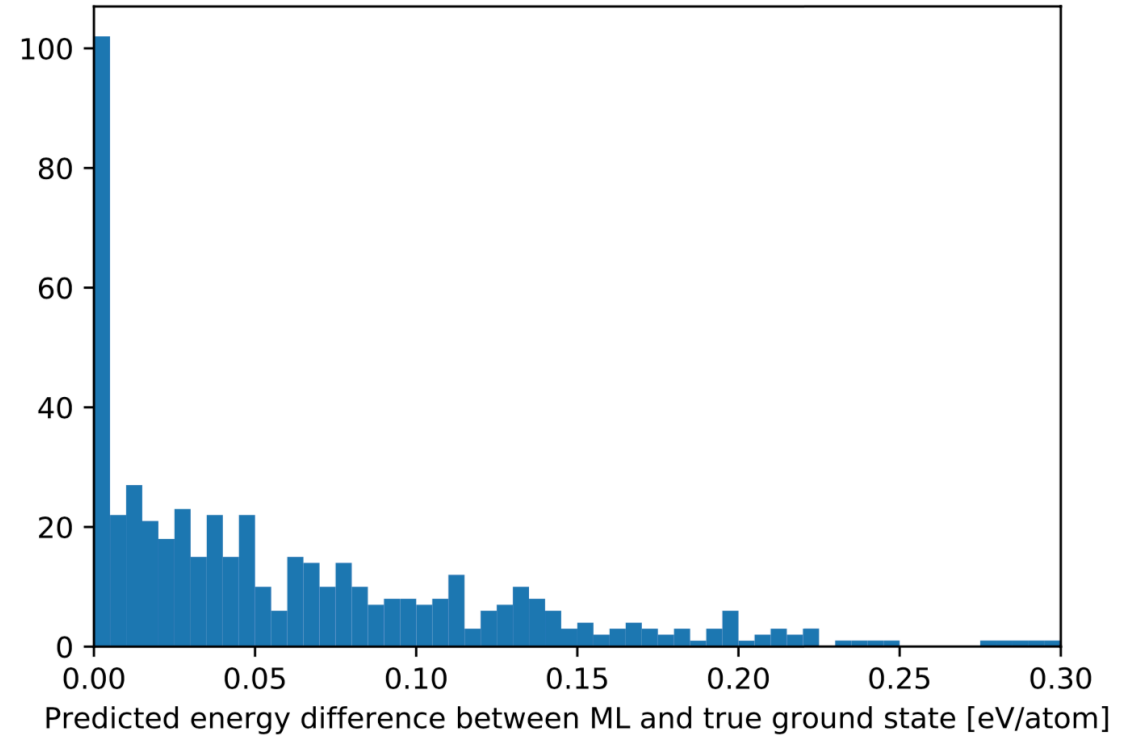
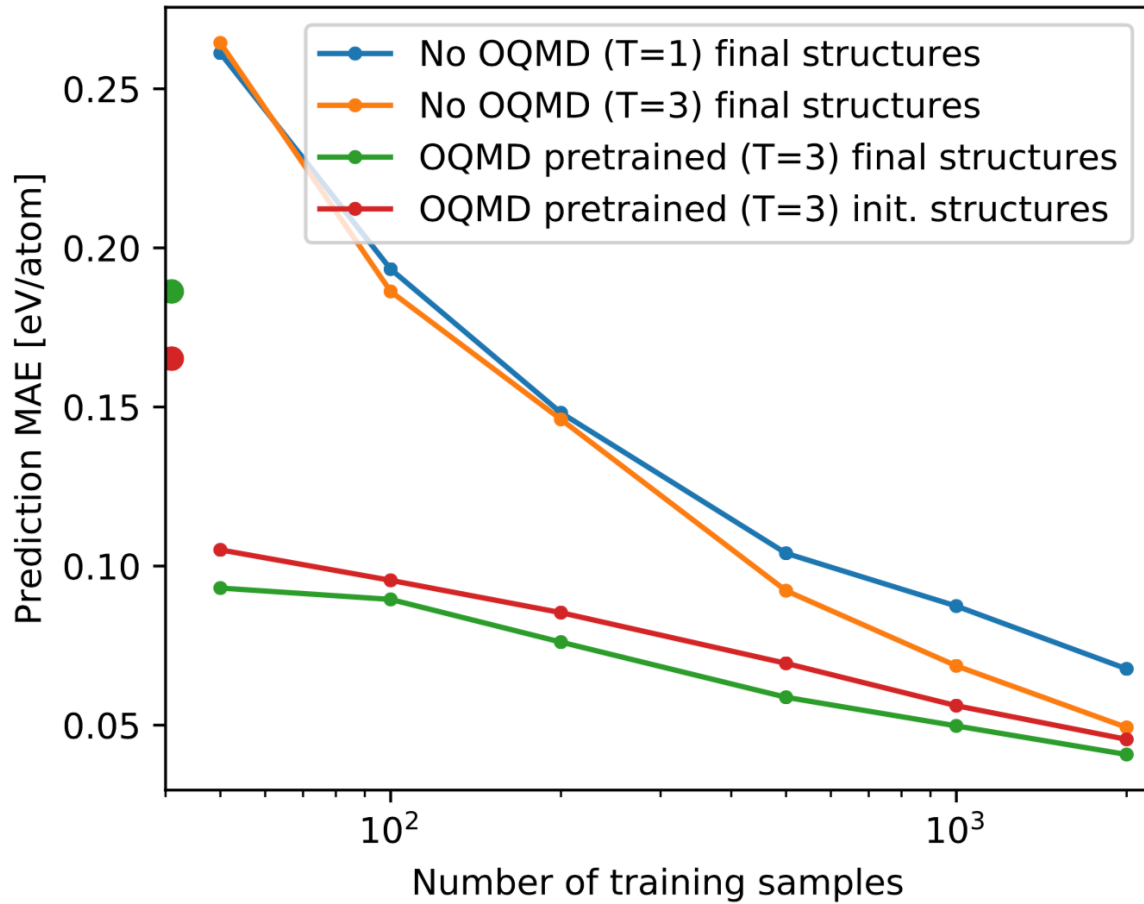


Additional training on 100 ABSe₃
MAE = 0.10 eV



ABSe₃ dataset

Additional training on ABSe₃ dataset



$$\Delta E = E^{ML}(G_{DFT}) - E^{ML}(G_{ML})$$

ML for computational screening

- Significant speed-up of materials screening with machine learning
- However severe limitations:
 - Better descriptors/understanding of solar cells/water splitting devices
 - Absorption
 - Defects
 - What limits the open-circuit voltage V_{oc} ?
 - More accurate calculations beyond DFT (GW/RPA/BSE)

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Mikkel N. Schmidt

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Brian Seger
Søren Dahl
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